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# Translating Excited State Dynamics to Classical Interatomic Potentials

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Sandia National Laboratories, Albuquerque, NM*

LAMMPS Workshop, Albuquerque, 2025



U.S. DEPARTMENT  
of ENERGY

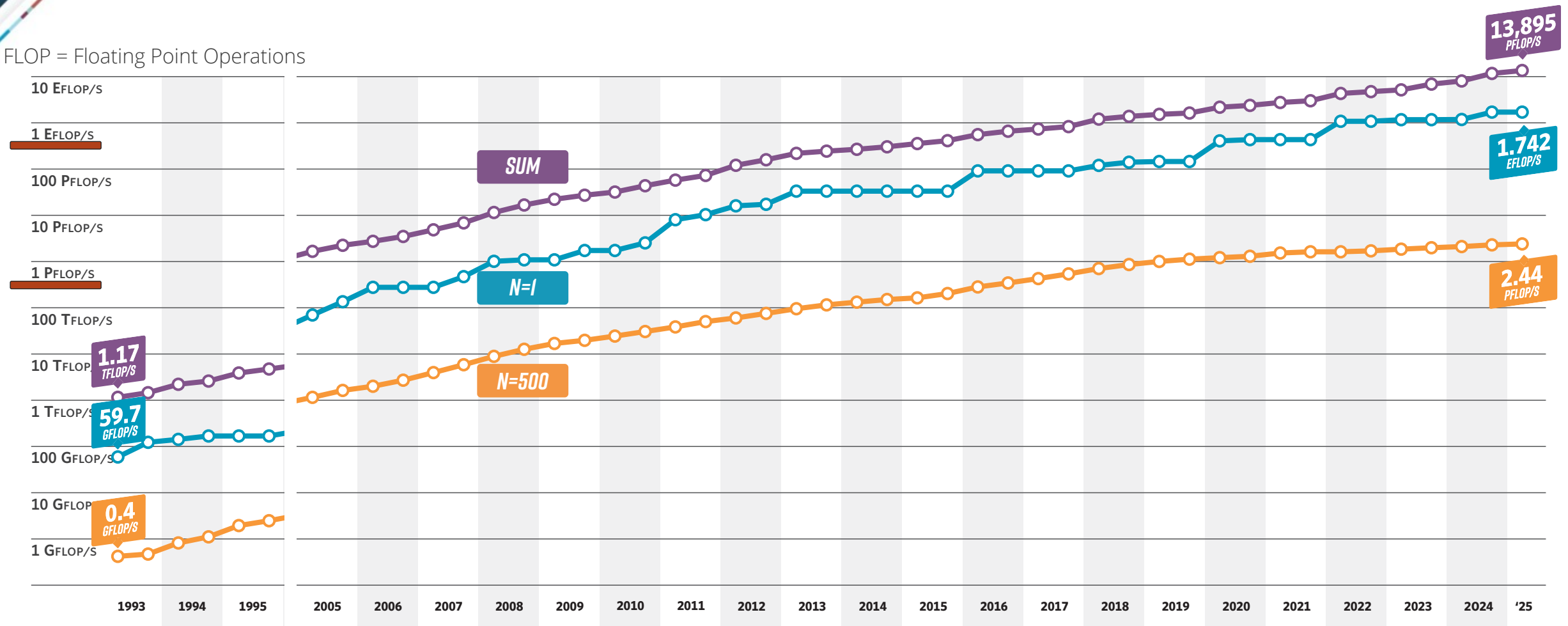


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# 21<sup>st</sup> Century Theorist (?)

FLOP = Floating Point Operations



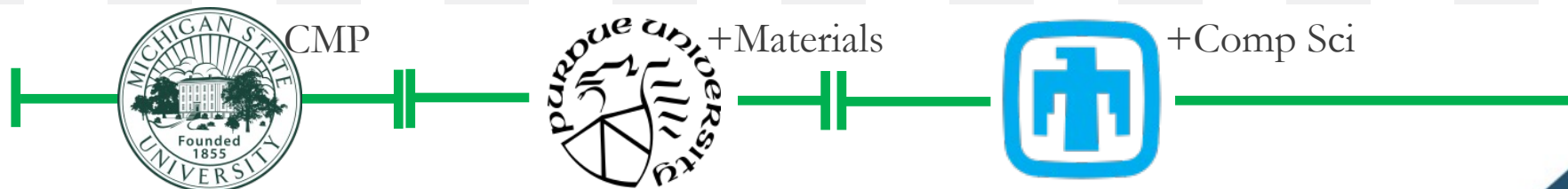
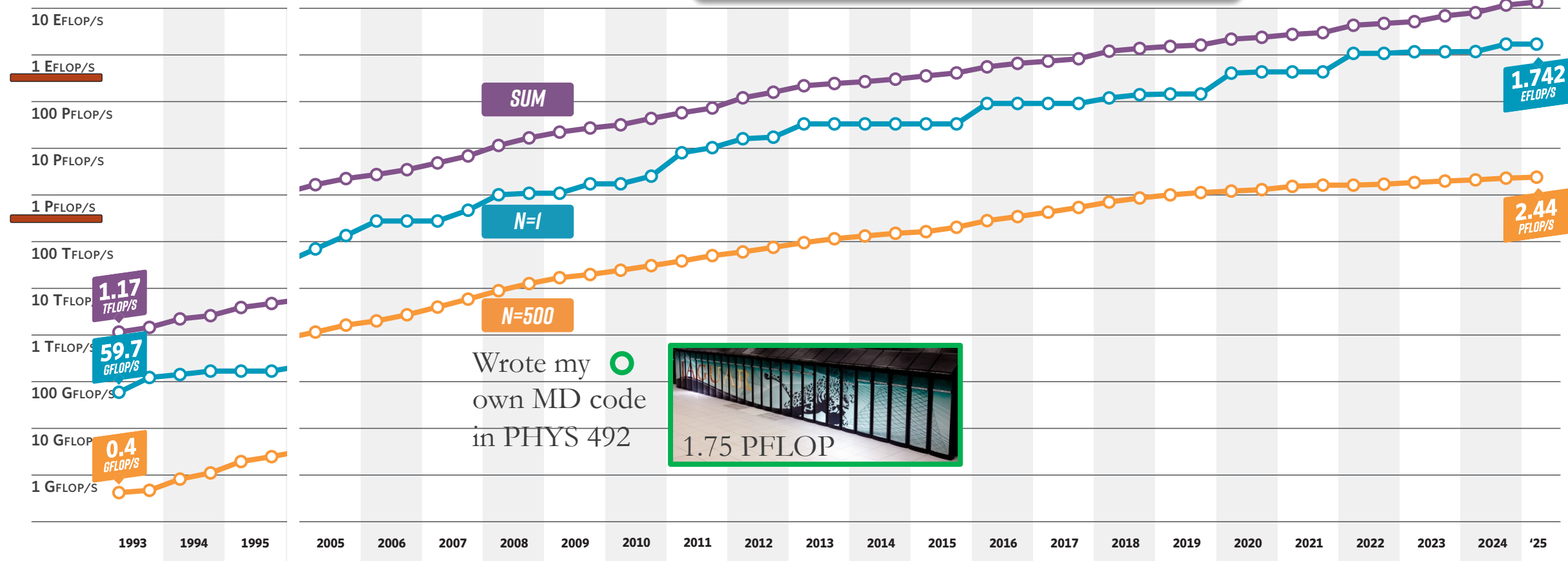
More powerful supercomputers are inevitable, but is our scientific usage of this technology keeping up?



# My Perspective

Entering Ugrad a PS5 would have been  
~400<sup>th</sup> most powerful supercomputer

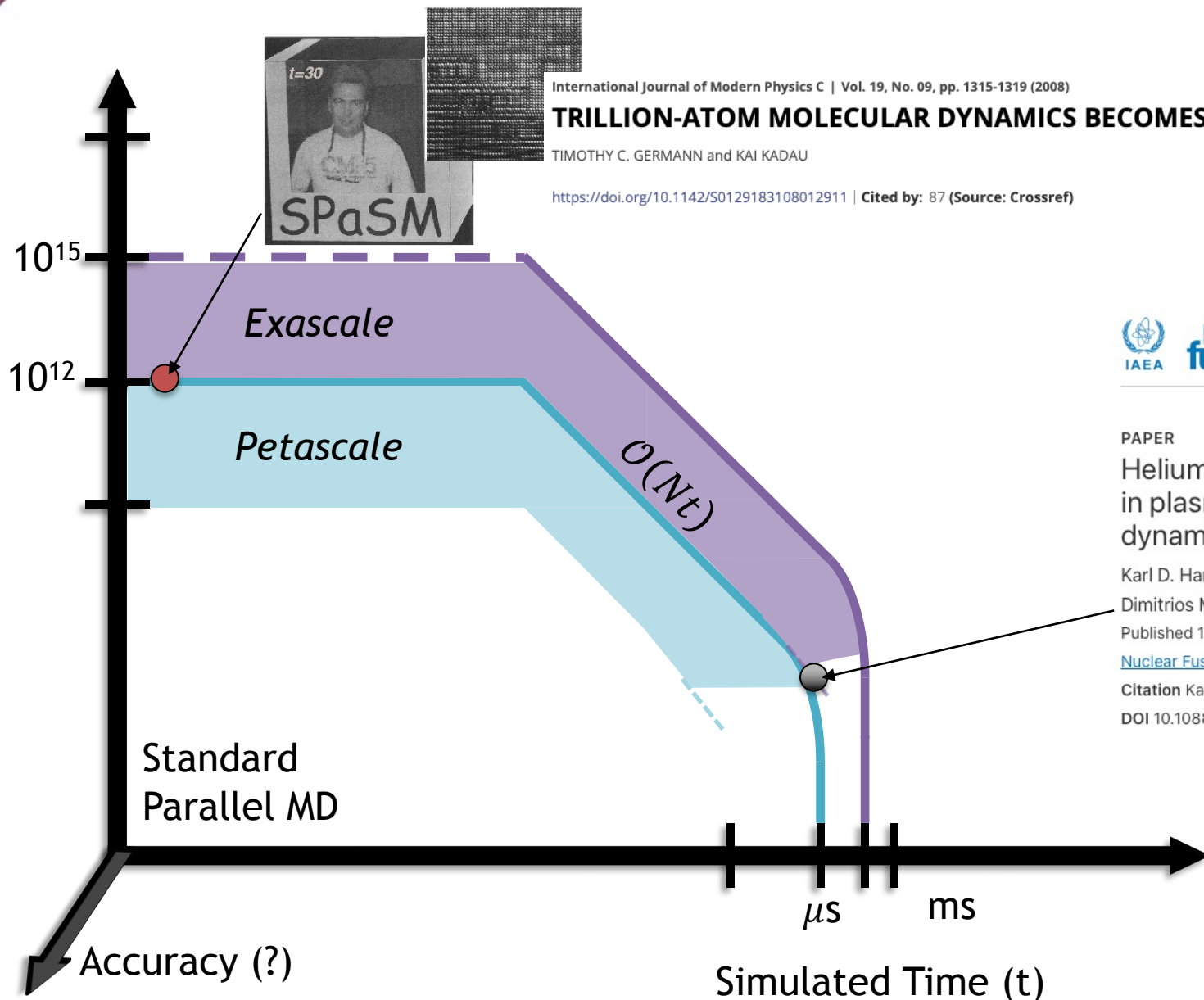
FLOP = Floating Point Operations





# Frontiers of Molecular Dynamics

Number of Particles (N)



International Journal of Modern Physics C | Vol. 19, No. 09, pp. 1315-1319 (2008)

## TRILLION-ATOM MOLECULAR DYNAMICS BECOMES A REALITY

TIMOTHY C. GERMANN and KAI KADAU

<https://doi.org/10.1142/S0129183108012911> | Cited by: 87 (Source: Crossref)



PAPER

Helium flux effects on bubble growth and surface morphology in plasma-facing tungsten from large-scale molecular dynamics simulations

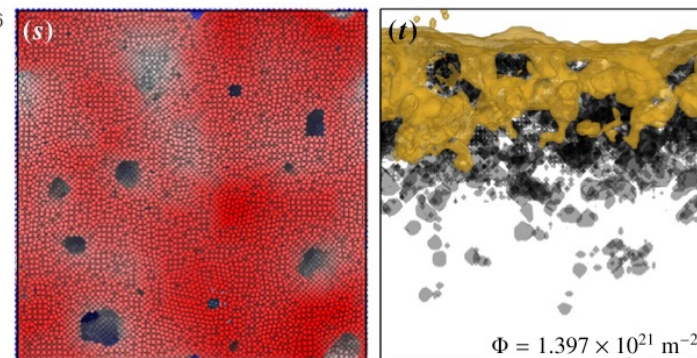
Karl D. Hammond<sup>1,2</sup> , Ian V. Naeger<sup>1</sup>, Wathsala Widanagamaachchi<sup>8,3</sup>, Li-Ta Lo<sup>4</sup>, Dimitrios Maroudas<sup>5</sup> and Brian D. Wirth<sup>6,7</sup>

Published 17 May 2019 • © 2019 IAEA, Vienna

[Nuclear Fusion](#), Volume 59, Number 6

Citation Karl D. Hammond et al 2019 *Nucl. Fusion* 59 066035

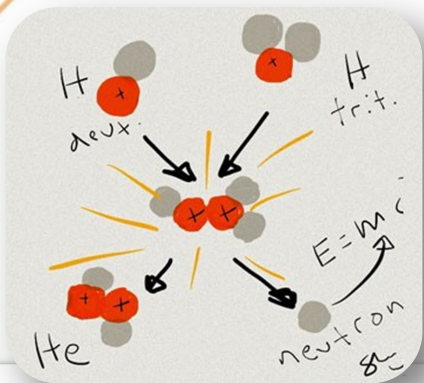
DOI 10.1088/1741-4326/ab12f6







# Exascale Molecular Dynamics



**To operate safely and efficiently**, fusion power plants must be designed with materials that can withstand the extreme heat and particle loads generated by the fusion process.

**Challenge:** Current approaches to material selection are too slow and expensive, because they rely on human physical intuition, domain expertise, and trial and error. Enter ML and Molecular Dynamics to help out.

**LAMMPS**

[gitlab.com/exascalematerialsproject/exascalematerialsproject](https://gitlab.com/exascalematerialsproject/exascalematerialsproject)  
[github.com/lammps/lammps](https://github.com/lammps/lammps)

**Outcome:** Demonstrated fast virtual assessment of new materials, from the training of accurate machine-learned models to large-scale, long-time exascale simulations

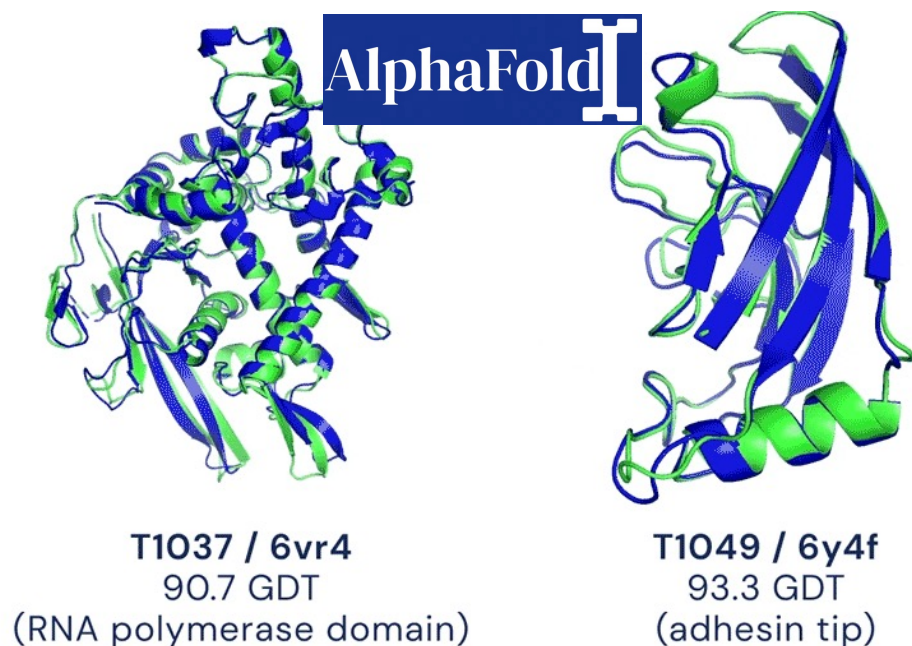
Captured surface erosion rates in a candidate material utilizing **4 Frontier-Days**.



# Exascale Molecular Dynamics

Observation 1: Machine learning has broken the traditional accuracy and cost tradeoff

Observation 2: Model development has been democratized, inviting broader uses of modeling / simulation methods



- Experimental result
- Computational prediction

Claim: Exascale computing power will be used to push the frontiers of simulation accuracy not length/time



What is a machine learned Interatomic potential?



A machine-learned interatomic potential refers to a computational model or algorithm that is designed to describe the interactions between atoms in a material using machine learning



What is the TLDR version of this I can use to explain this at an undergraduate level



Machine-learned interatomic potentials are computational models that use machine learning techniques to describe how atoms interact in materials. In traditional approaches, equations and parameters are used to approximate these interactions. However, machine-learned potentials take a different approach by training on large datasets to learn the relationship between atomic structures and their properties.





# MD Approximations Change Over Time

<http://lammps.sandia.gov>

## Twobody (B.C.)

Lennard-Jones, Hard  
Sphere, Coulomb, Bonded

## Manybody (1980s)

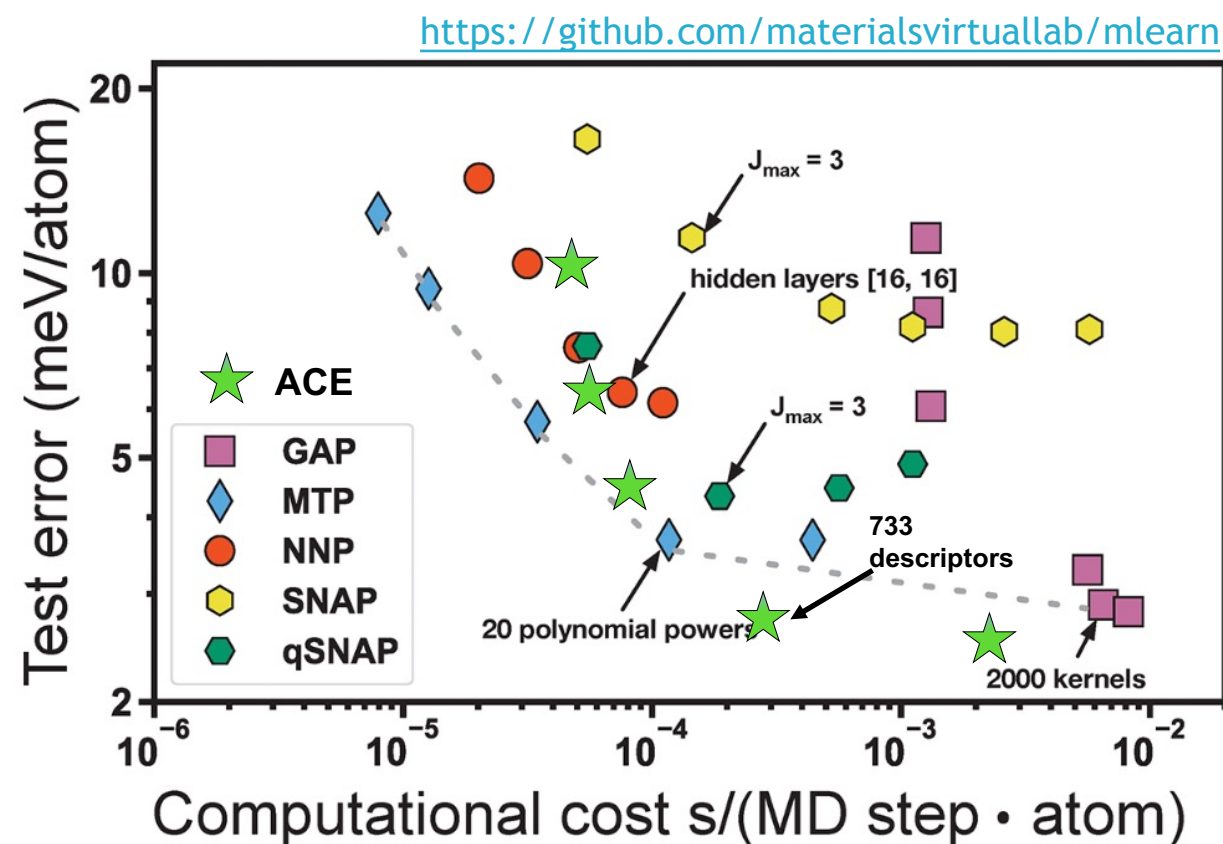
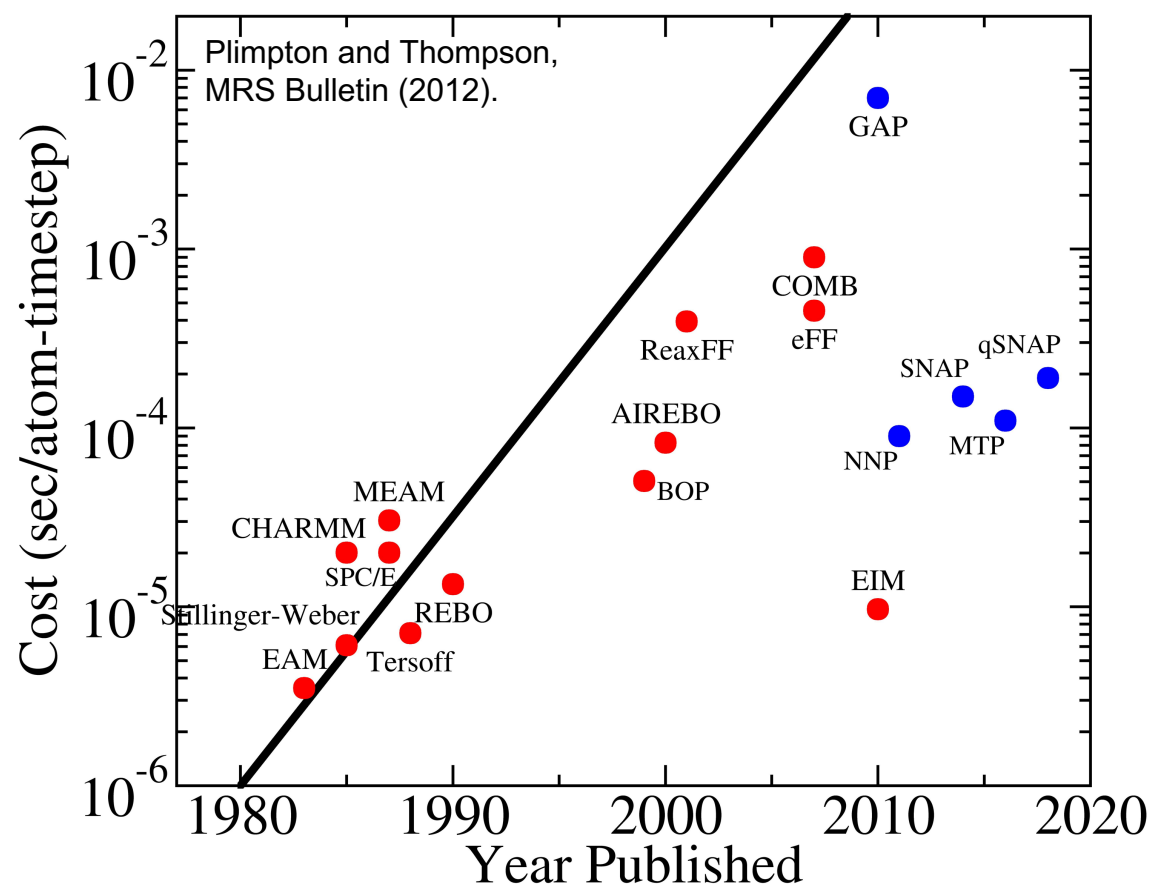
Stillinger-Weber, Tersoff,  
Embedded Atom Method

## Advanced (90s-2000s)

REBO, BOP, COMB,  
ReaxFF

## Big Data / Deep / Machine Learning (2010s)

GAP, SNAP, NN,...

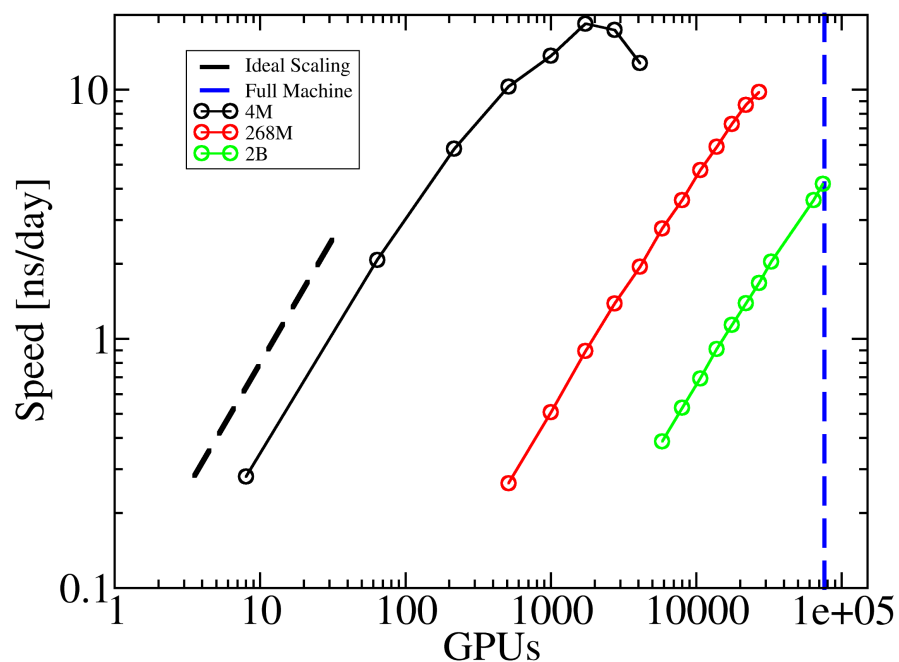




# Exascale Ready Models

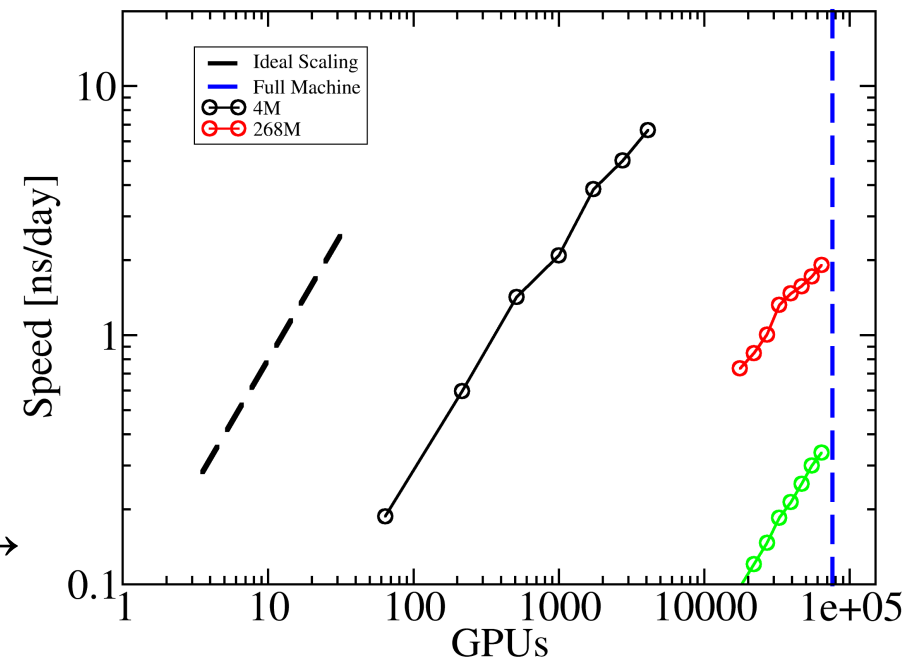
## Performance Metrics

- Different descriptors, different computational cost.
- LAMMPS with ML-IAP are now a qualification metric for new exascale platforms!



← Bispectrum Components

Atomic Cluster Expansion →

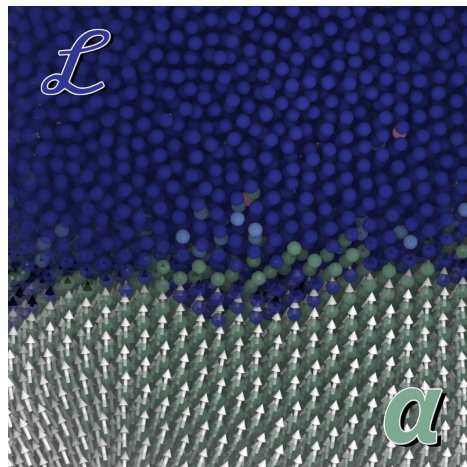




# What is Next Gen MD?

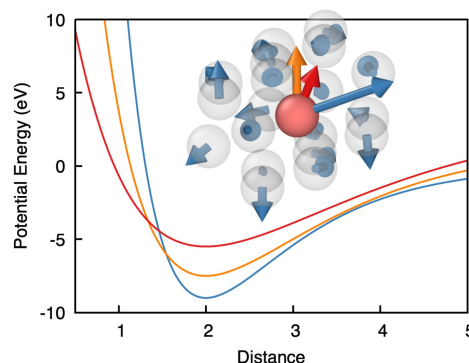
## Magnetic Effects

- Available in LAMMPS, but developments needed for many-body spin interactions and other novel Hamiltonians



## Electronic Excitations

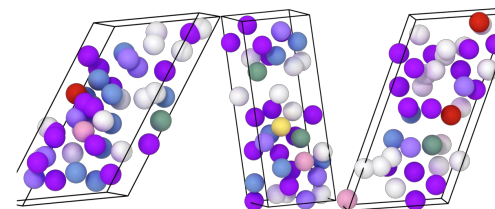
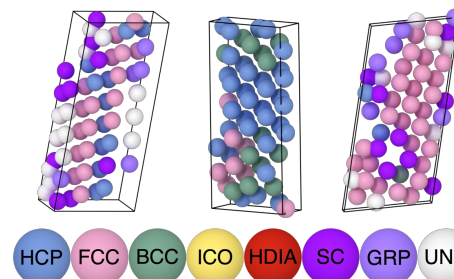
- Aspirational, but achievable given the rapid advancements of ML-IAP



## Novel Descriptor Usage

- What is the value added beyond accuracy of using a ML-IAP given the computational cost?

- $K > 0$  : Minimize Local Entropy



- $K < 0$  : Maximize Local Entropy

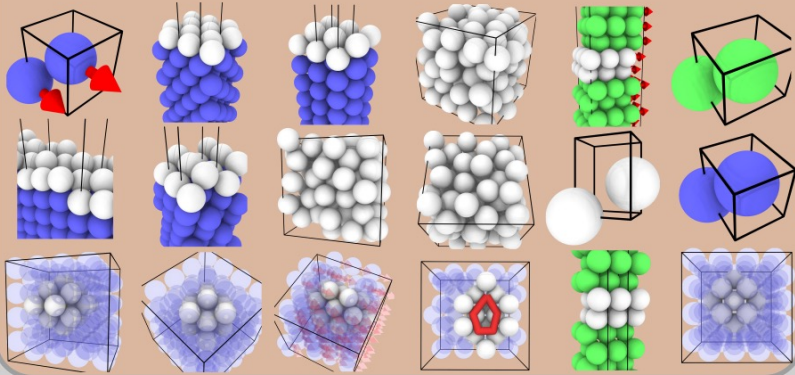
**Assembled  
Training**



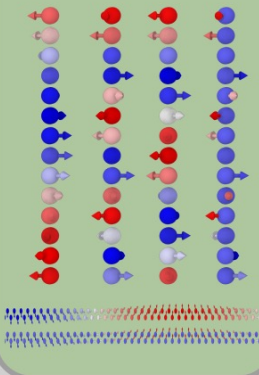
# Multi-Physics Molecular Dynamics

## First-Principles Training Set

### DFT Calculations

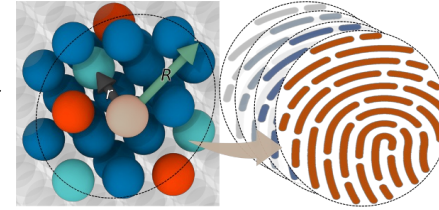


### Spin Spirals



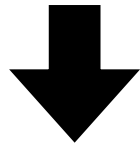
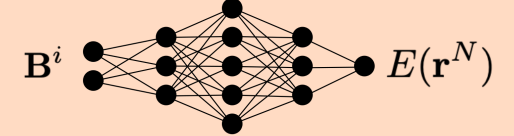
## Atomic Interaction

$$\sum_i \frac{\mathbf{p}_i^2}{2m} + E(\mathbf{R})$$



## SNAP Models

$$\beta \cdot \mathbf{B}^i + \frac{1}{2} (\mathbf{B}^i)^T \cdot \alpha \cdot \mathbf{B}^i = E(\mathbf{r}^N)$$



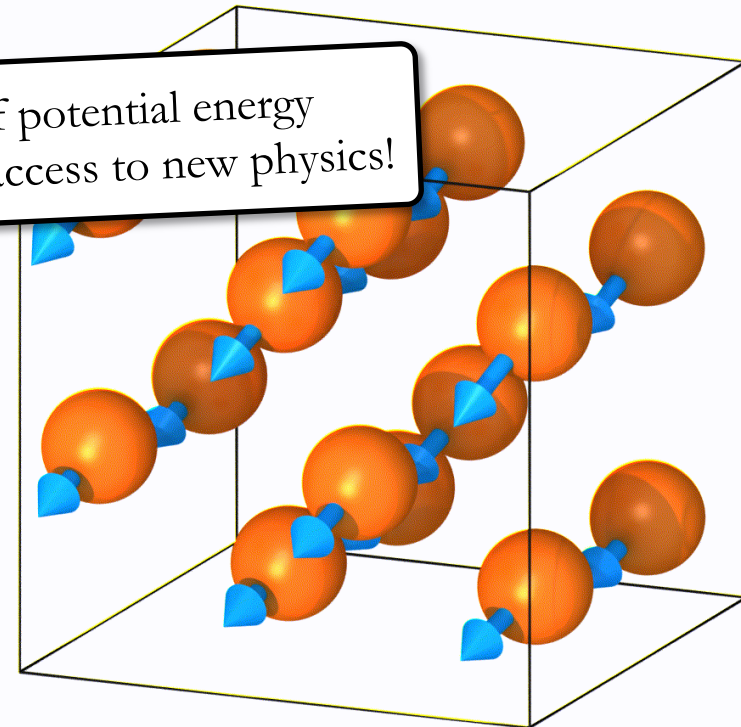
## Spin Exchange Interaction

$$\sum_{i,j}^N J_{ij}(\mathbf{R}) [\vec{s}_i \cdot \vec{s}_j - 1] - \sum_{i,j}^N K_{ij}(\mathbf{R}) [(\vec{s}_i \cdot \vec{s}_j)^2 - 1]$$

## Spin-Crystal Anisotropy Interaction

$$\frac{1}{2} \sum_{i,j=1, i \neq j}^N \left\{ \begin{aligned} & l_1(\mathbf{R}) \left[ (\mathbf{e}_{ij} \cdot \mathbf{s}_i)(\mathbf{e}_{ij} \cdot \mathbf{s}_j) - \frac{\mathbf{s}_i \cdot \mathbf{s}_j}{3} \right] + \\ & q_1(\mathbf{R}) \left[ (\mathbf{e}_{ij} \cdot \mathbf{s}_i)^2 - \frac{\mathbf{s}_i \cdot \mathbf{s}_j}{3} \right] \left[ (\mathbf{e}_{ij} \cdot \mathbf{s}_j)^2 - \frac{\mathbf{s}_i \cdot \mathbf{s}_j}{3} \right] \\ & + q_2(\mathbf{R}) \left[ (\mathbf{e}_{ij} \cdot \mathbf{s}_i)(\mathbf{e}_{ij} \cdot \mathbf{s}_j)^3 + (\mathbf{e}_{ij} \cdot \mathbf{s}_j)(\mathbf{e}_{ij} \cdot \mathbf{s}_i)^3 \right] \end{aligned} \right\}$$

Layering of potential energy surfaces gives access to new physics!



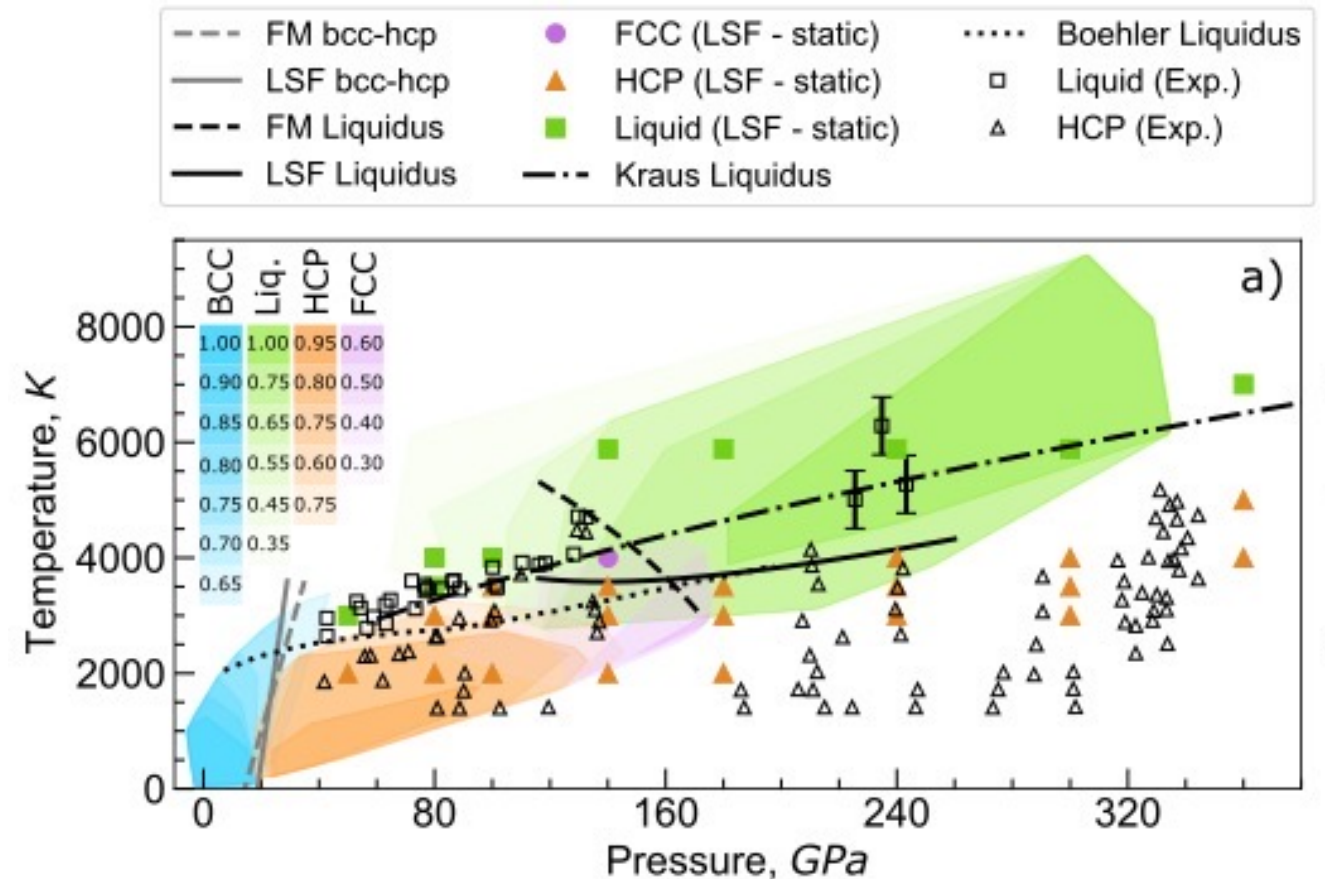
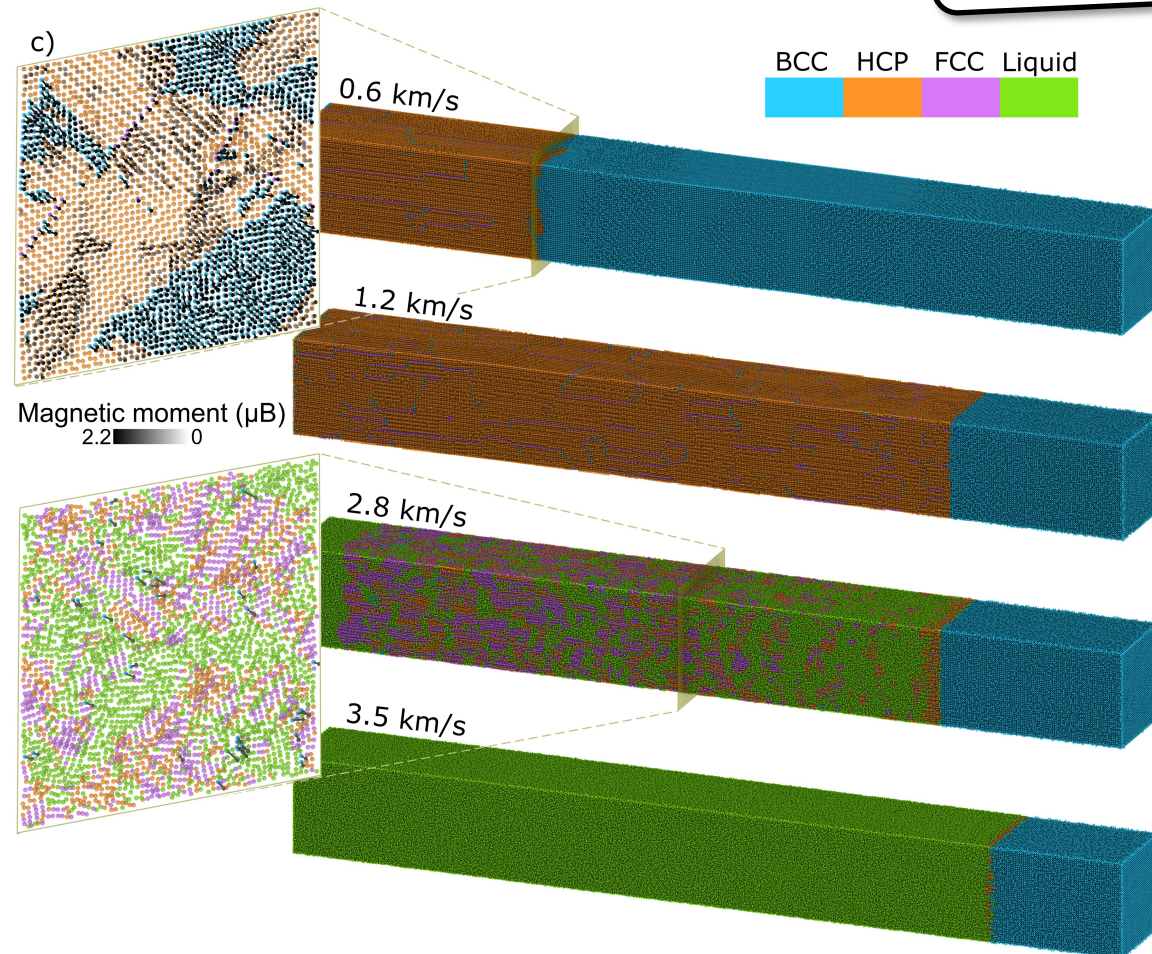




# Atomic Insight Into the Earths' Core

“Probing iron in Earth’s core with molecular-spin dynamics”  
Nikolov, S ... Wood, M. PNAS 2024

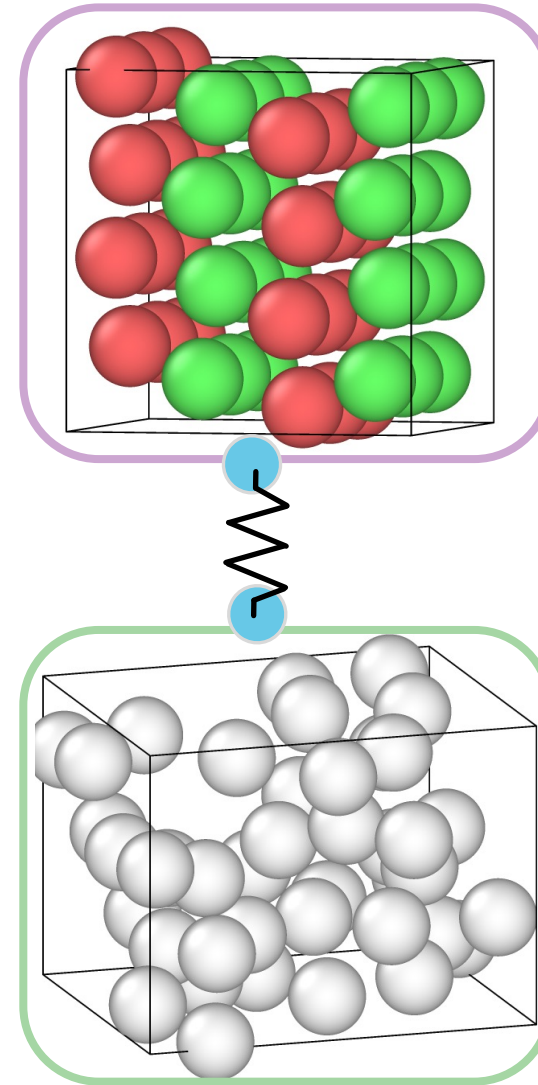
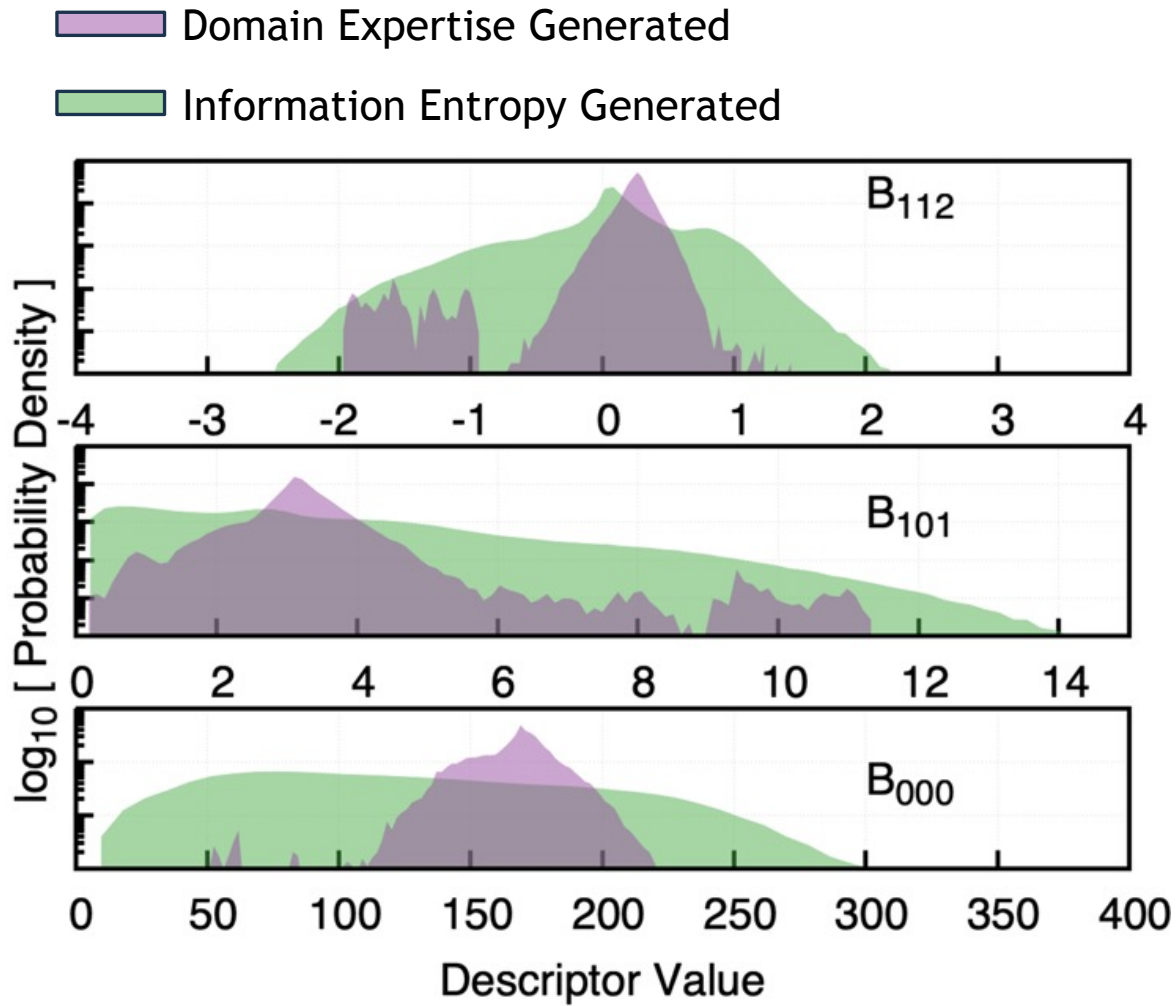
Magnetic and structural transitions captured  
in one MD simulation → Efficient sampling  
of dynamic phase diagram!





# Value Added in MD From Atomic Descriptors

- Data compression easily achieved on large simulations by reducing MD trajectory into descriptor distributions

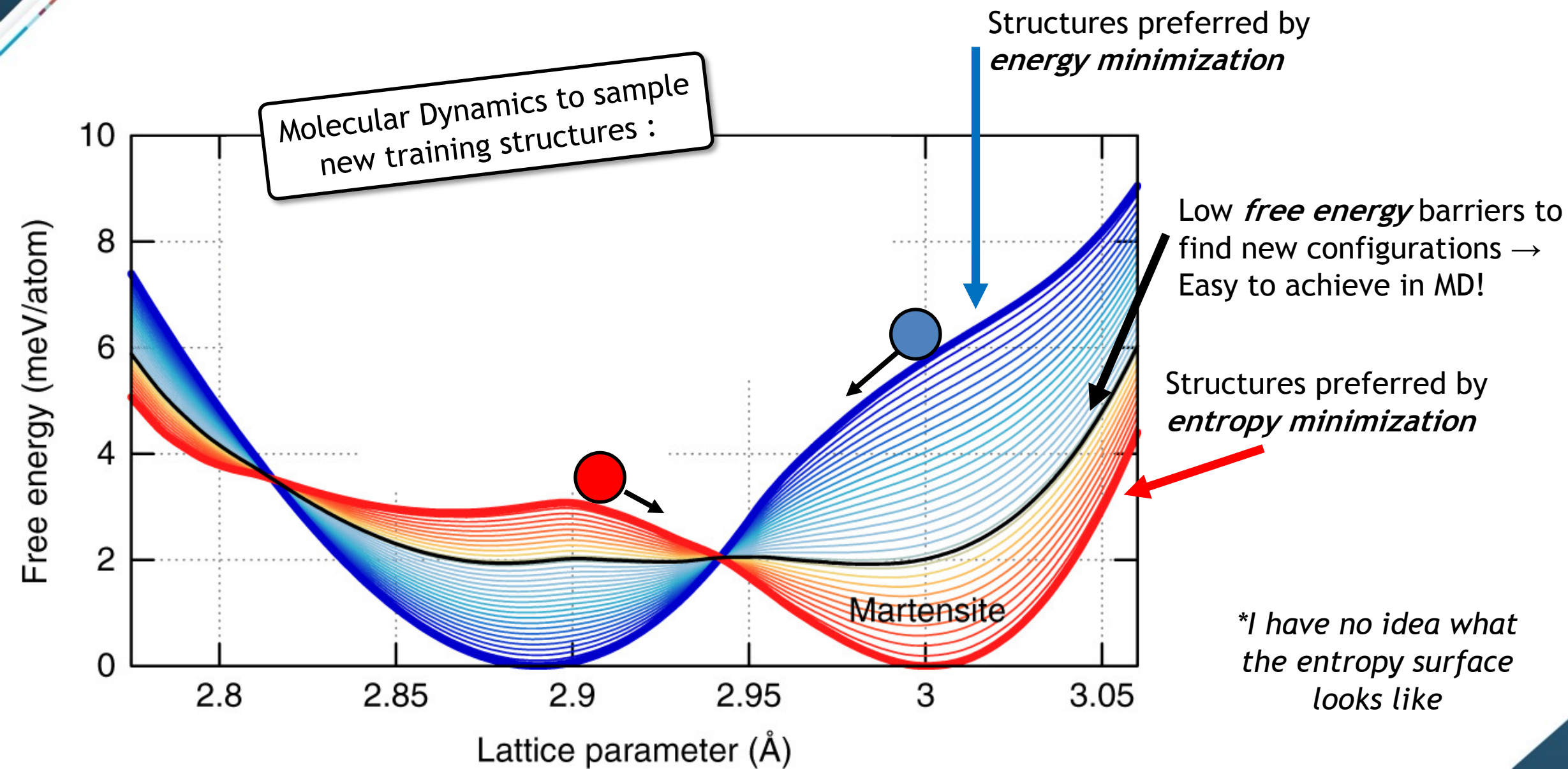


- Descriptors can be used as a scale agnostic collective variable
- Information entropy on the descriptor distribution can be used to apply a force on atoms to move toward/away known structures





# Value Added in MD From Atomic Descriptors







# Generalized Representative Structures

"Generalized representative structures  
for atomistic systems"

OPEN ACCESS

IOP Publishing

Journal of Physics: Condensed Matter

J. Phys.: Condens. Matter **37** (2025) 075901 (13pp) <https://doi.org/10.1088/1361-648X/ad9791>

- Each atom is represented by a vector of  $\sim 100$  descriptors, any structure represented by distributions of descriptor values.

Score Target Proposed

$$Q_{\text{SQS}} = \sum_{\alpha} |\bar{\Phi}_{\alpha}(\sigma) - \bar{\Phi}_{\alpha}^{\text{rand}}|$$

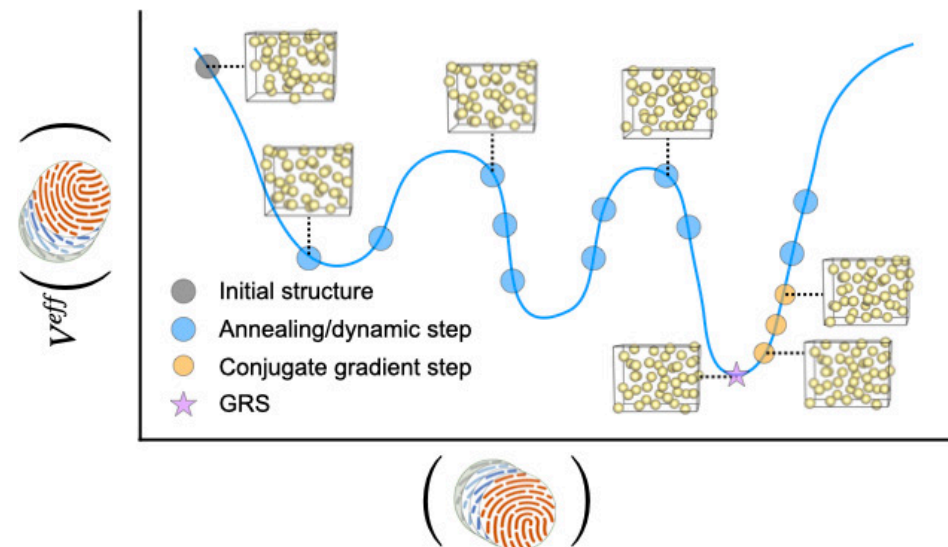
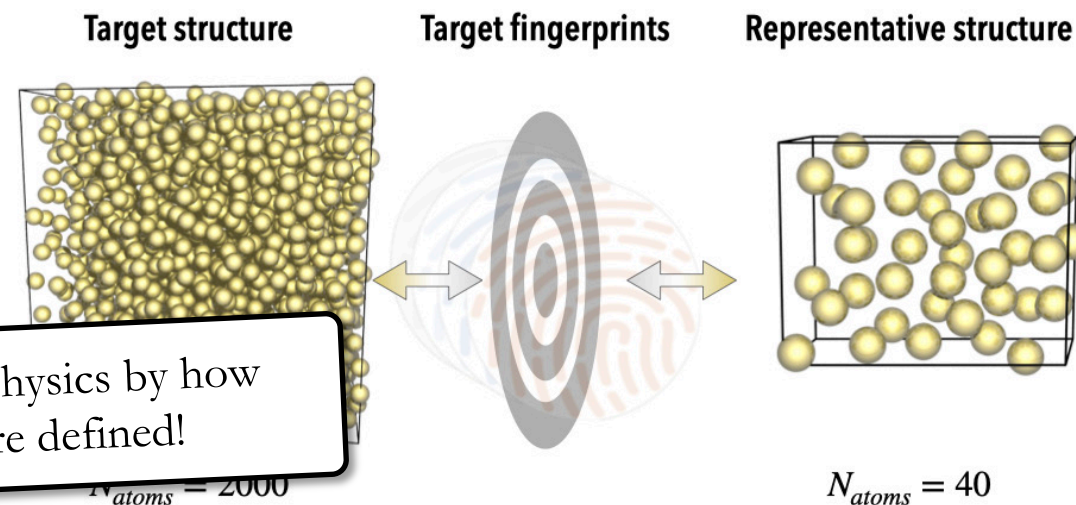
Access to new physics by how  
potentials are defined!

- Can move between scales by simplifying the descriptors,  $d^s(\{\Phi_{\alpha}\})$  and  $d^t(\{\Phi_{\alpha}\})$  with their moments

$$Q_{\text{GRS}}^{\text{intra}} = \alpha_1 \Delta \mathcal{M}_1[d^s, d^t] + \alpha_2 \Delta \mathcal{M}_2[d^s, d^t] + \dots$$

- Now construct an effective potential that is the difference between target and current structure, sampling is challenging.

$$V^{\text{eff}}(\{\mathbf{R}\}_S, \{\sigma\}_S) = V_{\text{core}}(\mathbf{R}, \sigma) + w_1 Q_{\text{GRS}}(\mathbf{R}_s, \sigma_s) + w_2 Q_{\text{GRS}}^{\text{intra}}(\{\mathbf{R}\}_S, \{\sigma\}_S).$$





# Dynamic Interatomic Potentials

## Approximate Electronic Excitations

- Even the most accurate ML-IAP fail to capture large changes in temperature due to electronic occupation changes.
- Born-Oppenheimer  $\rightarrow$  Sommerfeld Potentials

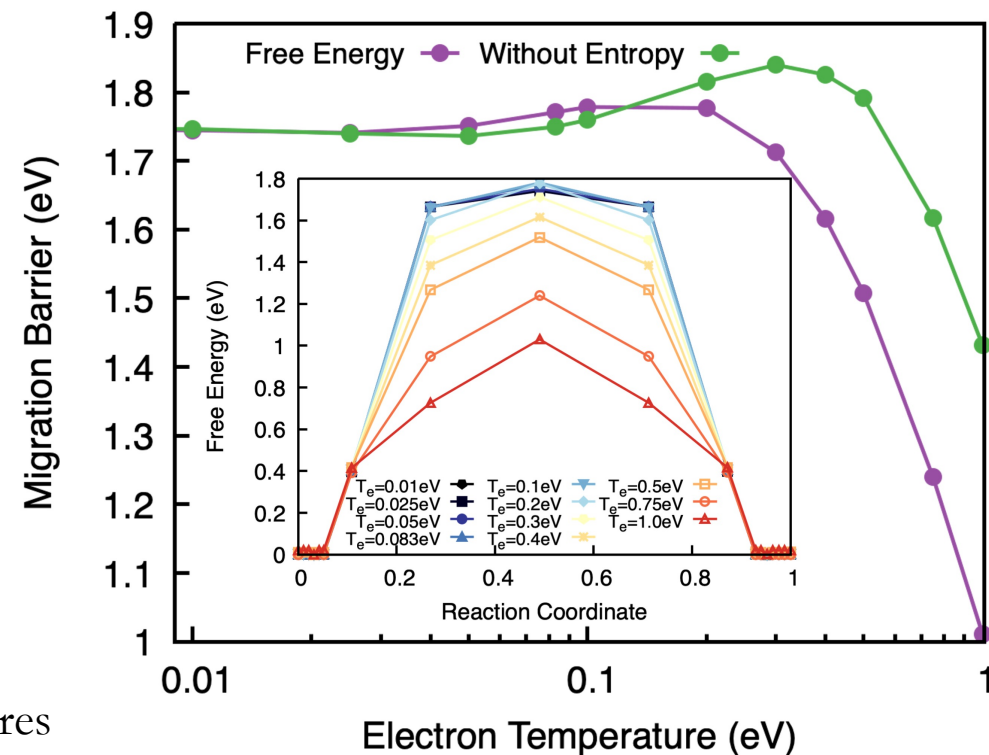
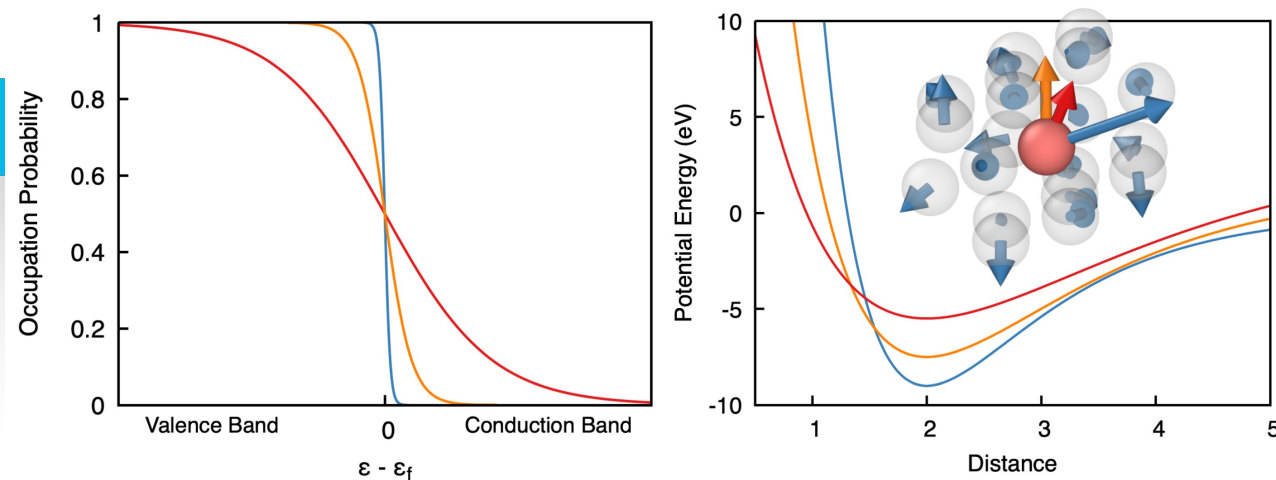
$$E = \int f_{FD}(\epsilon, T_e) D(\epsilon, \{R\}) d\epsilon \quad ; \quad f_{FD} = \frac{1}{e^{\epsilon/k_b T_e} + 1}$$

$$E = \int_{-\infty}^{\epsilon_f} \mathcal{H}(\epsilon) d\epsilon + \sum_n^{\infty} (k_b T)^{2n} a_n \frac{d^{2n-1}}{d\epsilon^{2n-1}} \mathcal{H}(\epsilon_f)$$

$$\mathcal{U}_{Som}(R_i) = \sum_j \mathcal{U}_{BO}(R_j, R_i) + T_e^2 \sum_j g_{\epsilon_f}(|R_i - R_j|)$$

Moment expansion for electron entropy can be captured with simple adaptations of ML-IAP

Pictorial PES Distortion by  $T_e$ :



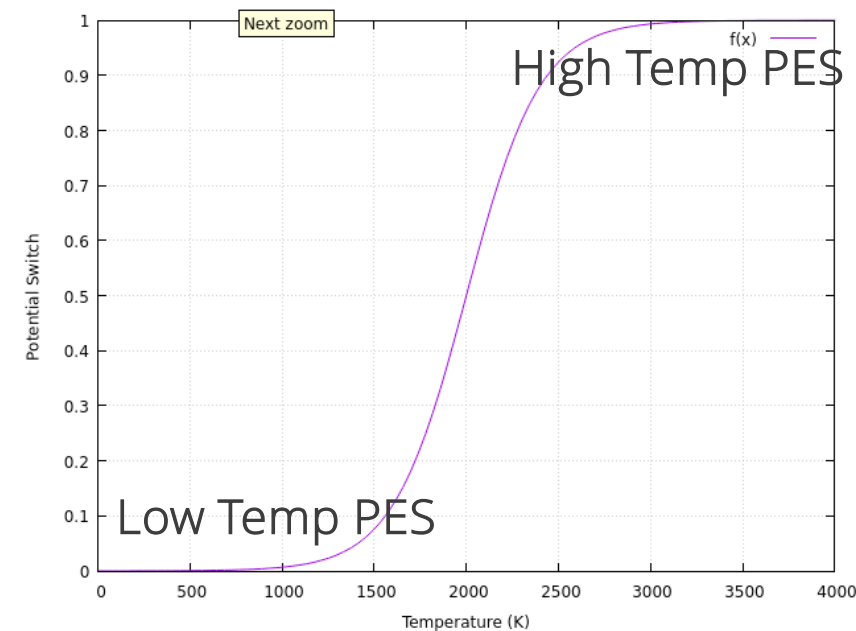
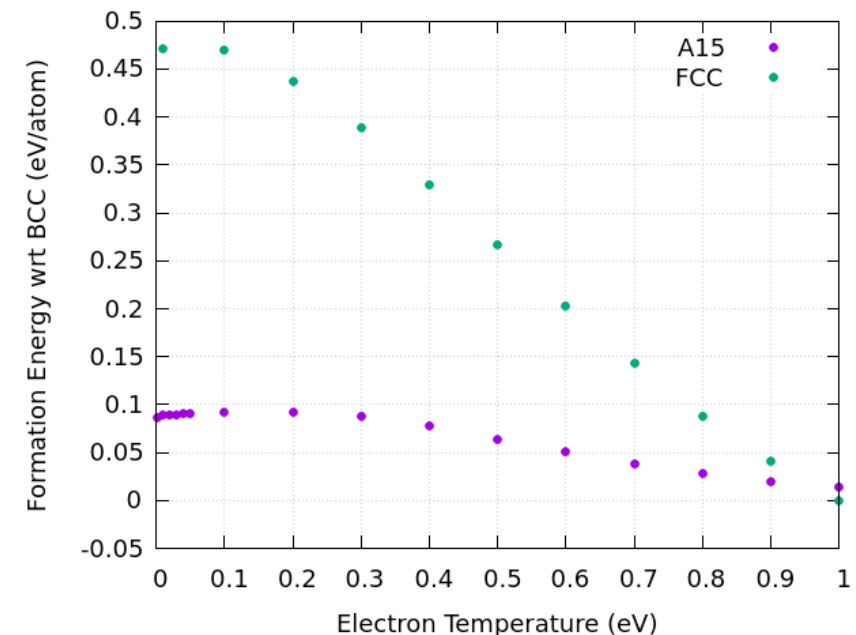
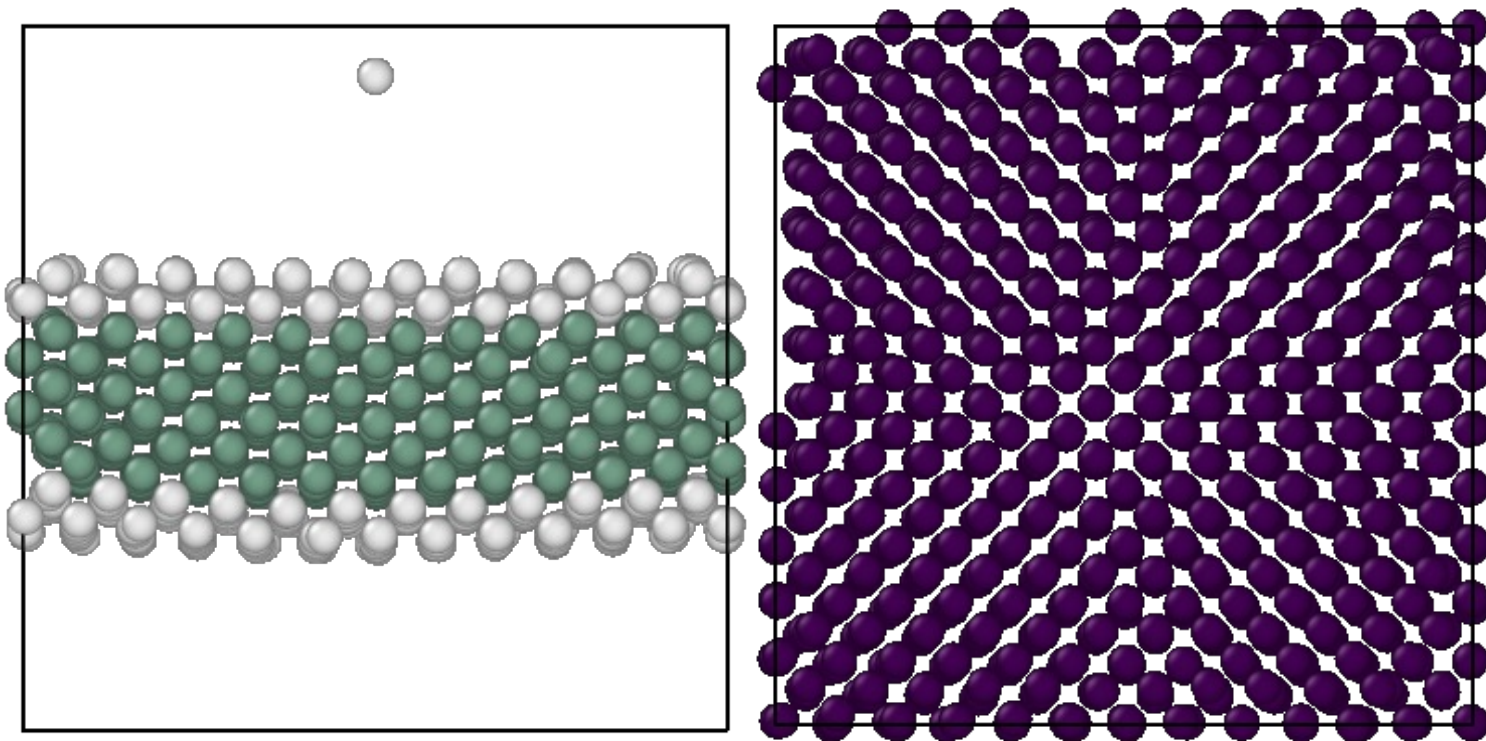
(Right) Vacancy migration barrier in Tungsten at elevated Temperatures



# In Practice Using LAMMPS

- Partition your training into two bins: High and Low Temperature

```
compute      lavg all ave/sphere/atom cutoff ${ravg}  
variable     psratio1 atom 1/(1+exp(-(c_lavg[2]-v_tdamp)/v_twidth)) #High T surface  
variable     psratio2 atom 1-v_psratio1                             #Low T surface  
pair_style    hybrid/scaled v_psratio2 eam/fs v_psratio1 pace product  
pair_coeff     * * eam/fs ../2013Marinica_W_EAM2.fs W  
pair_coeff     * * pace W_pot.yace W
```



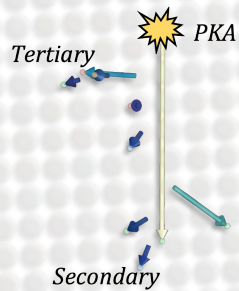




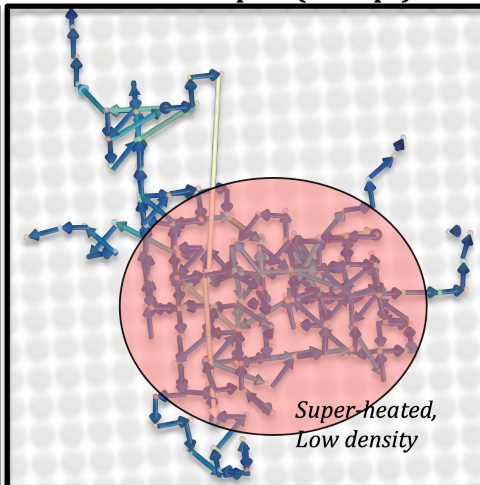
# Use Case for Radiation Damage

- During a neutron collision with the lattice, very high energy density states are generated
- Local temperature/density of atoms will determine which PES we sample from
- PKA = Primary Knock-on Atom

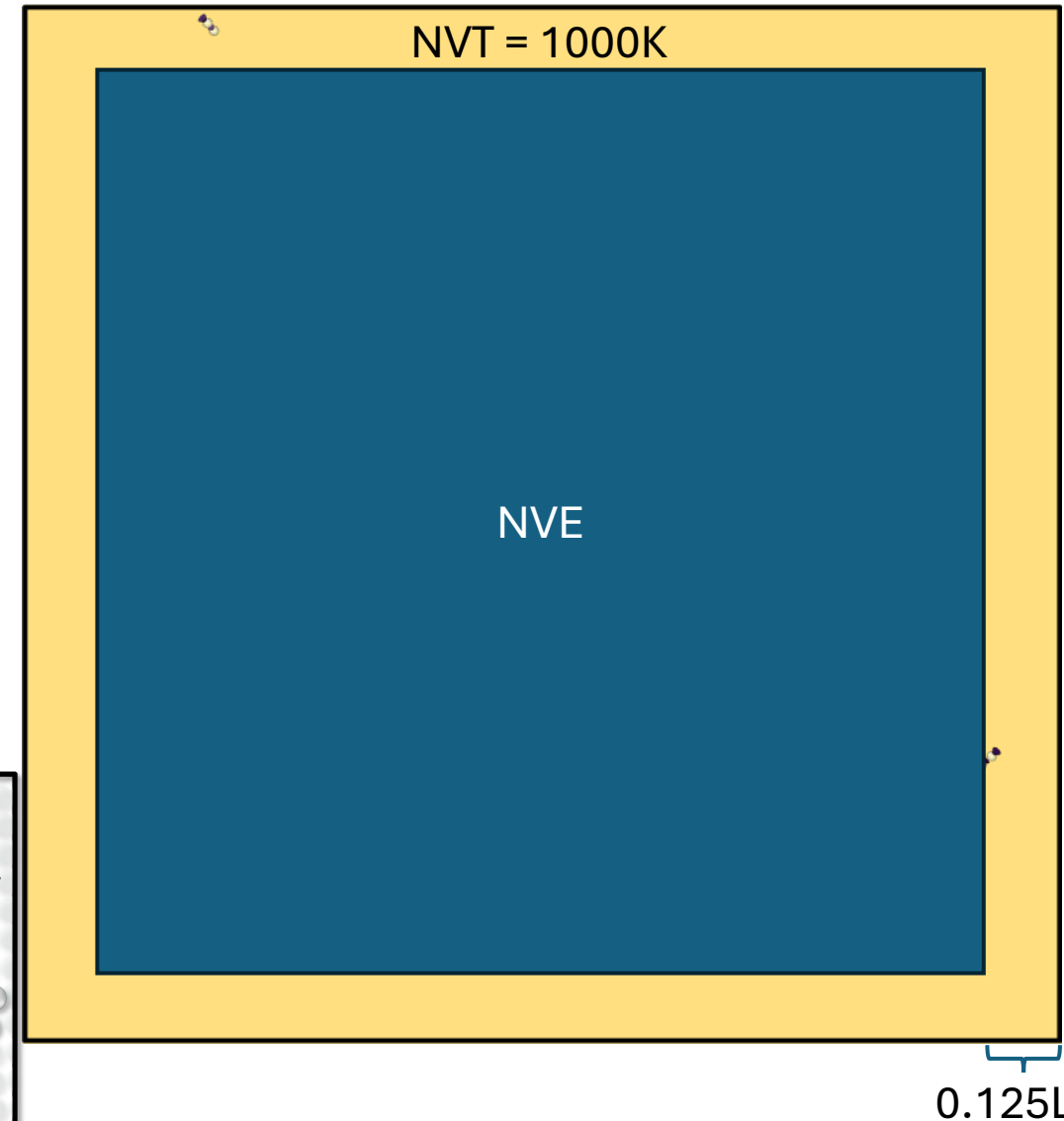
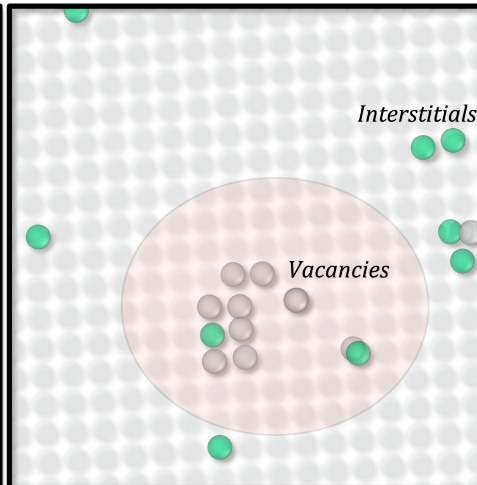
Initial Collision (~0.1ps)



Thermal Spike (1 - 10ps)



Equilibrium Evolution (100ps - )

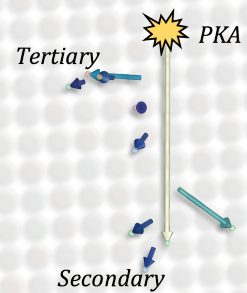




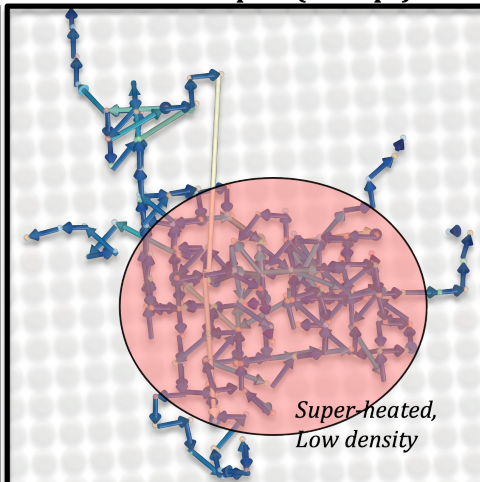
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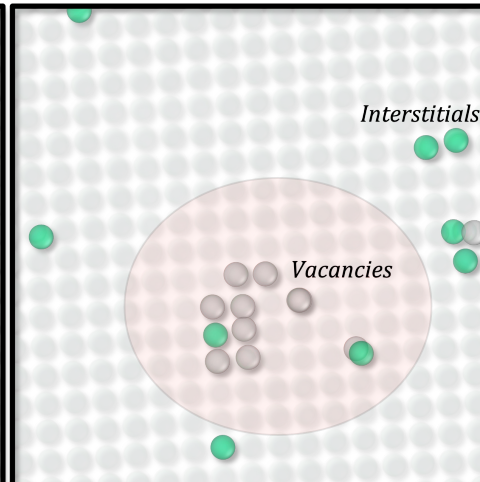
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Thermal Spike (1 - 10ps)

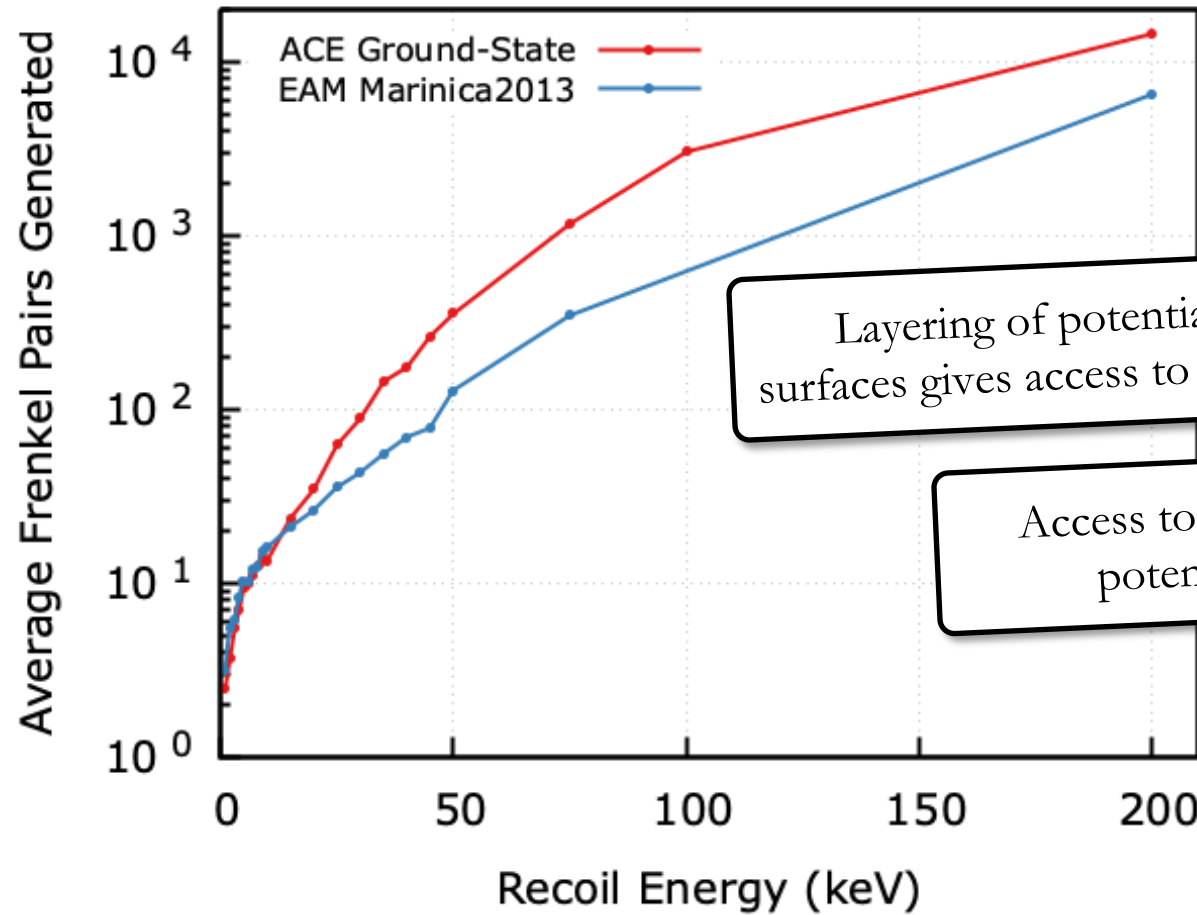


Equilibrium Evolution (100ps - )





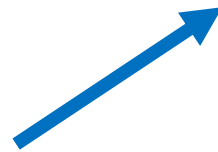
# Use Case for Radiation Damage



Layering of potential energy surfaces gives access to new physics!

Access to new physics by how potentials are defined!

ACE PES  $T=0.0\text{eV}$ , 100keV PKA  
7.3E-4 DPA before relax  
0.52ps Elapsed, adaptive timestep interval

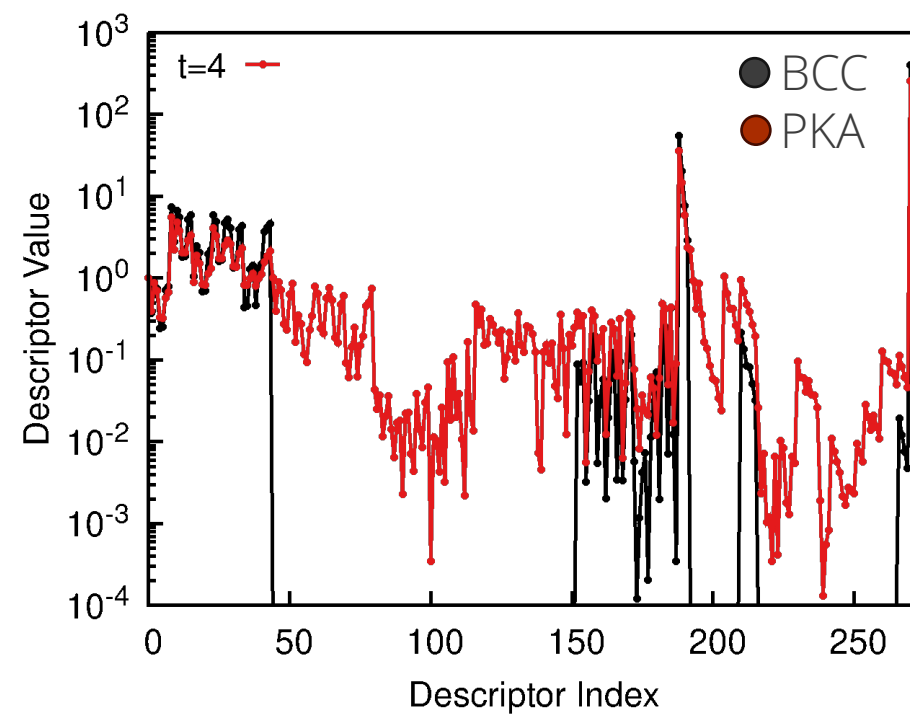
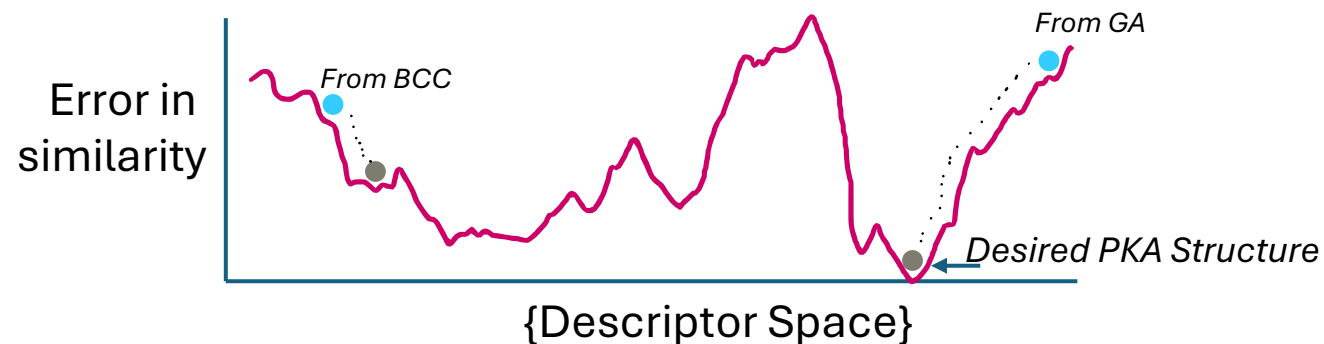
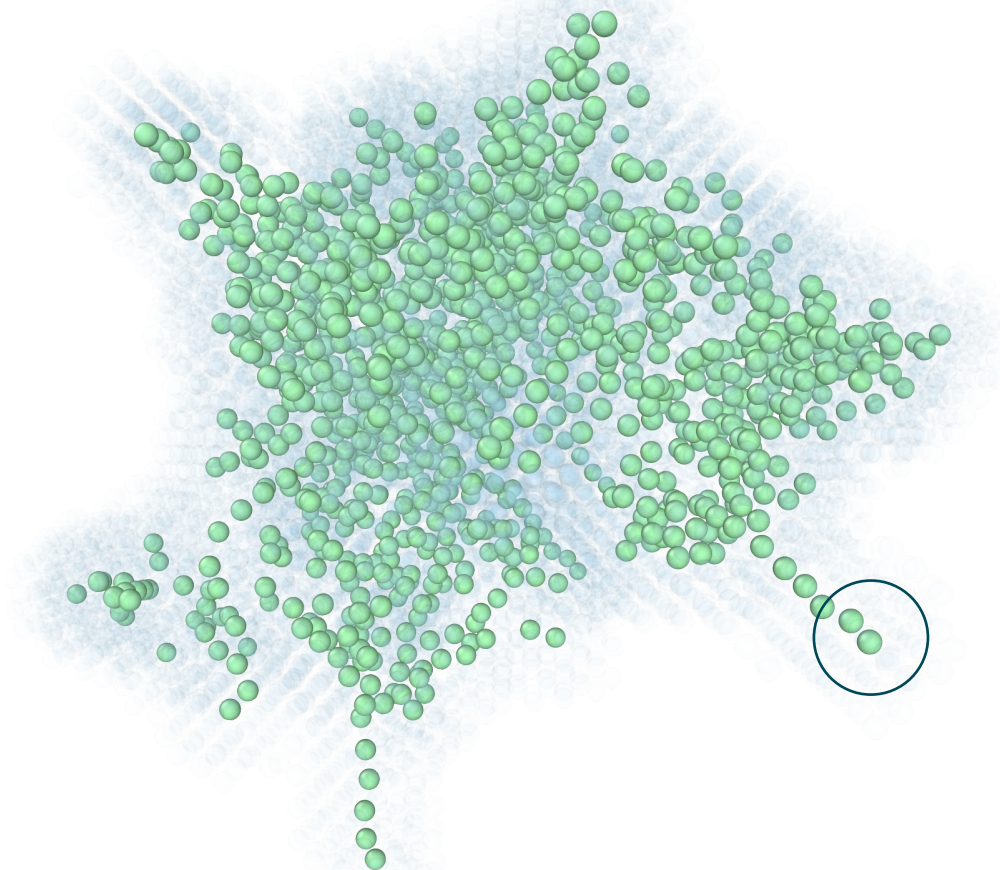






# Application to Radiation Damage in Materials

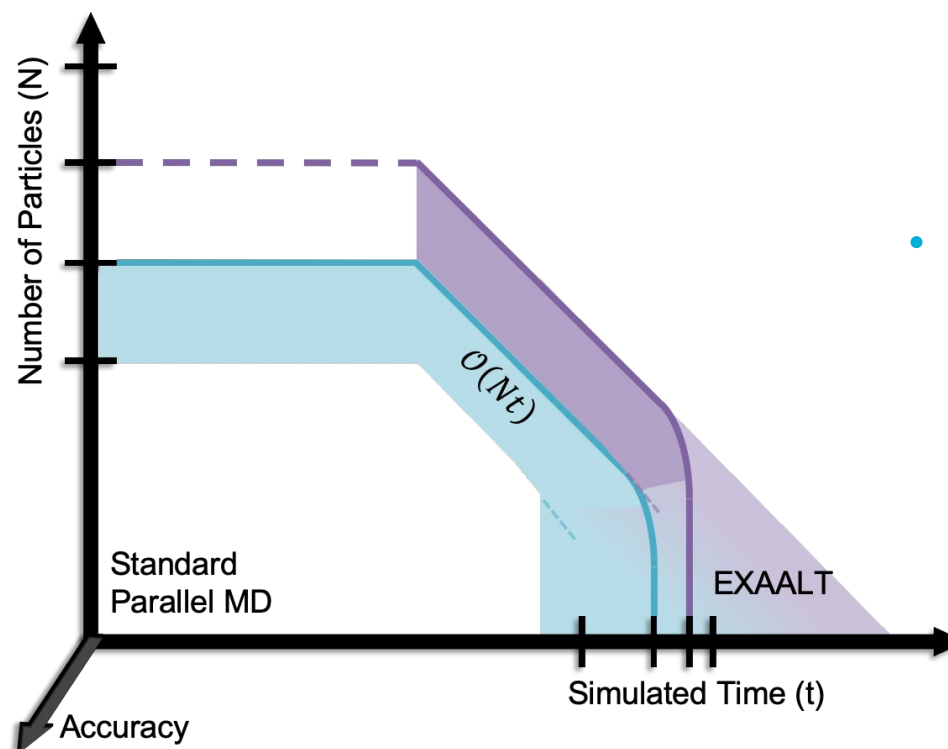
- ML-IAP are being refined toward accurate displacement damage by curating training to transient atomic structures





## Conclusion and Path Forward

- Data-driven interatomic potentials allow for MD predictions of challenging material problems.
- While harder to quantify, the fidelity of our MD simulations needs to be a key consideration



- Thank you to all my collaborators: Aidan Thompson, Mary Alice Cusentino, Krupa Ramasesha, Svetoslav Nikolov, Drew Rohskopf, Charlie Sievers, David Montes do Oca Zapian, Danny Perez, Nick Lubbers, Julien Tranchida, Steve Plimpton, Ivan Oleynik, Jon Willman, Ember Sikorski, Megan McCarthy, James Goff, James Stewart, Carlos Pereyra, Nat Trask, Michael Sakano, and many others!



Contact Information:  
[mitwood@sandia.gov](mailto:mitwood@sandia.gov)



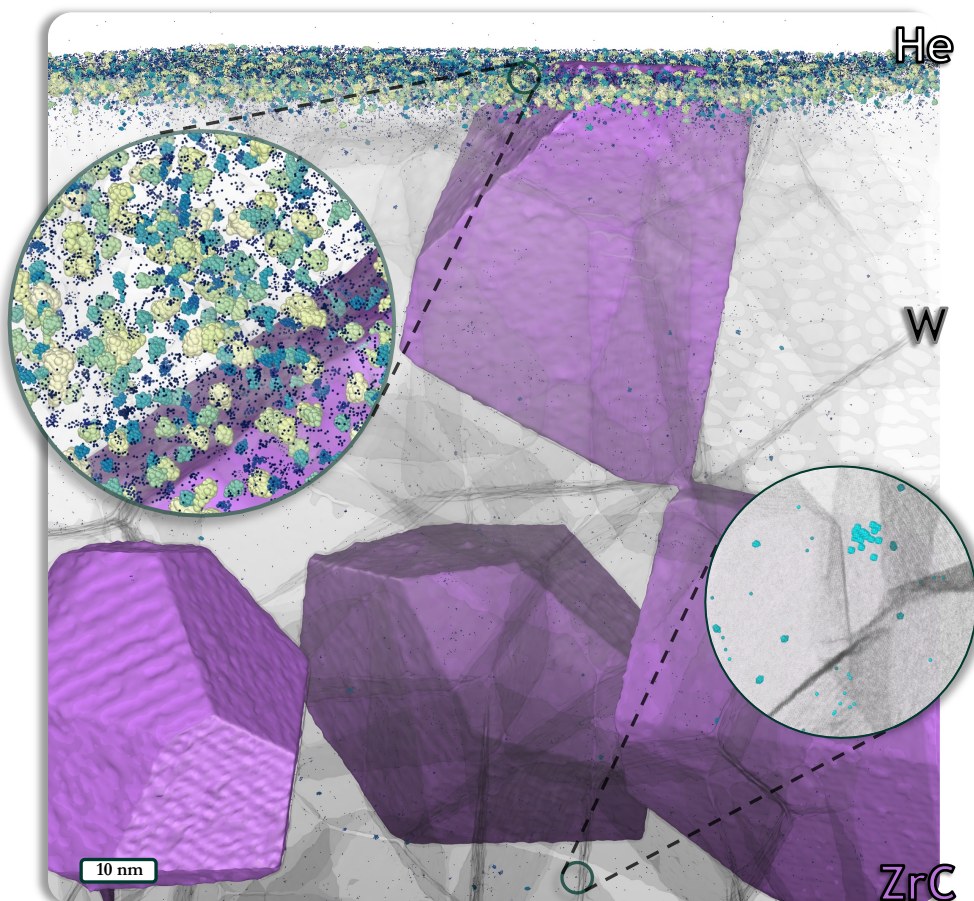
Office of  
Science





## Conclusion and Path Forward

- While harder to quantify, the fidelity of our ML-IAP training sets needs to be a key consideration



“Generalized representative structures  
for atomistic systems”

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IOP Publishing

Journal of Physics: Condensed Matter

J. Phys.: Condens. Matter **37** (2025) 075901 (13pp) <https://doi.org/10.1088/1361-648X/ad9791>

“Permutation-adapted complete and independent basis  
for atomic cluster expansion descriptors”

Journal of Computational Physics

Volume 510, 1 August 2024, 113073



FitSNAP /  
structure-generation

Contact Information:

[mitwood@sandia.gov](mailto:mitwood@sandia.gov)



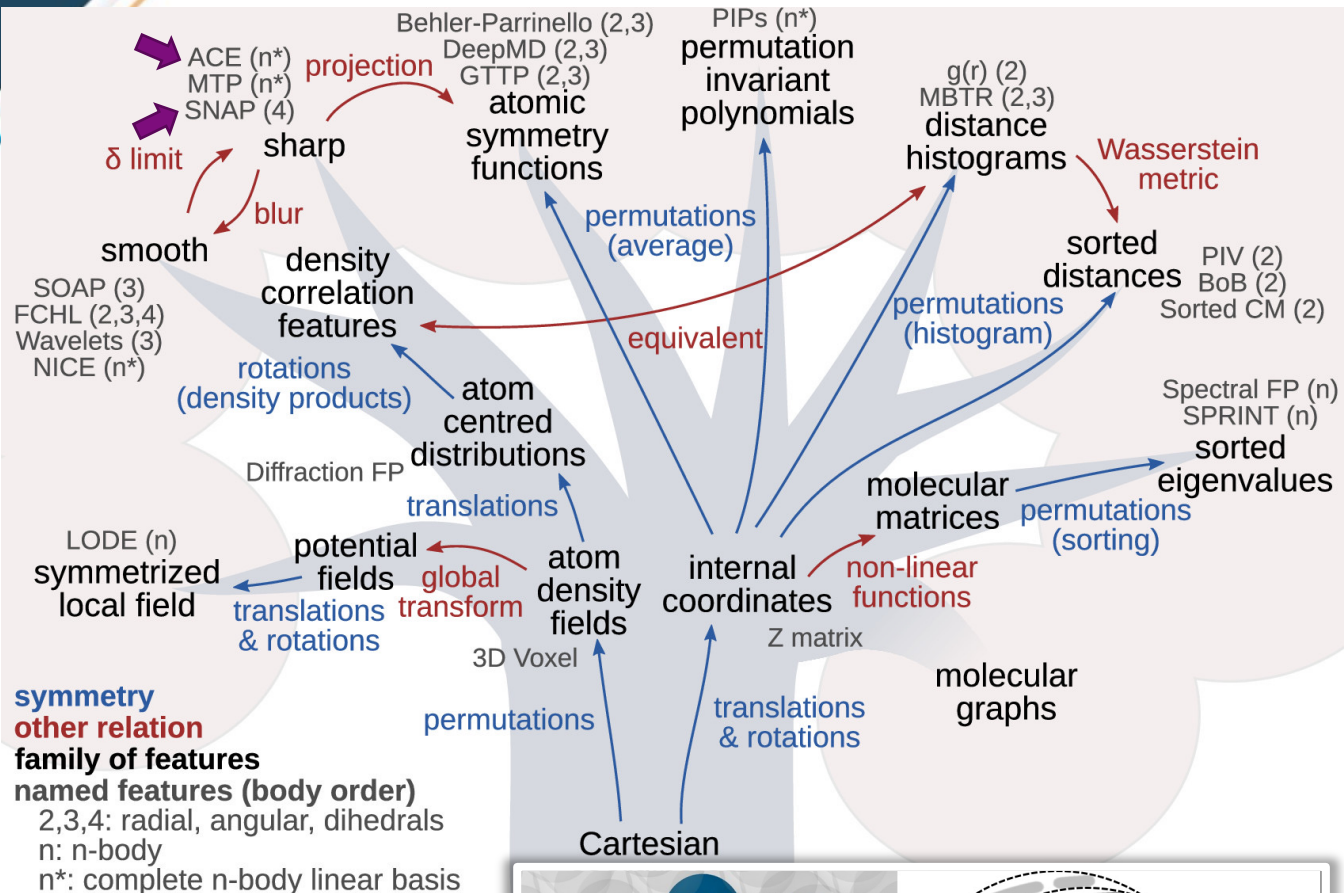
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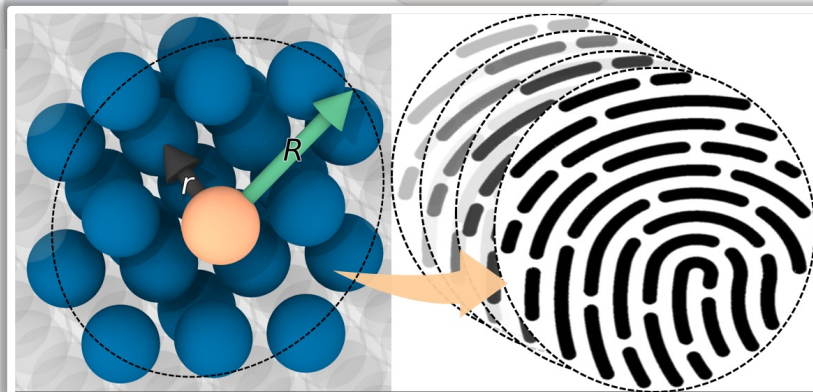


# Descriptor Sets



**symmetry**  
**other relation**  
**family of features**  
**named features (body order)**  
2,3,4: radial, angular, dihedrals  
 $n$ :  $n$ -body  
 $n^*$ : complete  $n$ -body linear basis

Musil *et. al.* Chem. Rev. (2021)  
121, 9759–9815



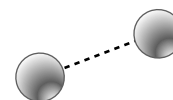
"Permutation-adapted complete and independent basis for atomic cluster expansion descriptors"

Journal of Computational Physics

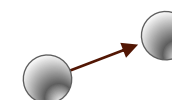
Volume 510, 1 August 2024, 113073

## Complete, generalizable single-bond basis

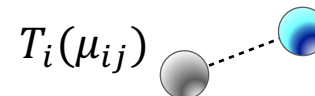
Spatial  
 $\phi_{inlm} = R_n(r_{ij})Y_l^m(\widehat{r_{ij}})$



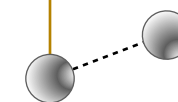
Charge (transfer)  
 $T_k(q_{ij})Y_l'^m(\delta\widehat{q_{ij}})$



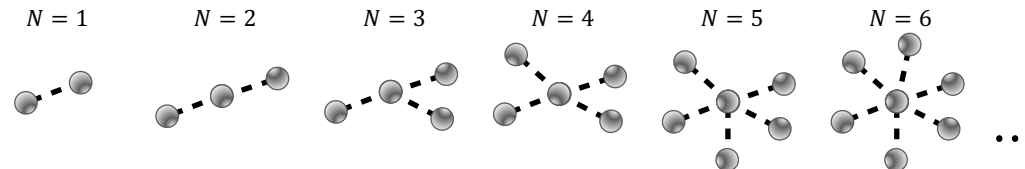
Chemical (bonds)



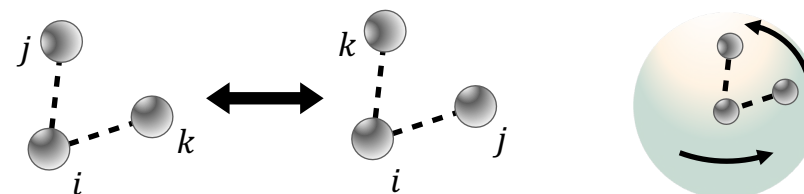
Magnetism  
 $T_k(M_i)Y_l''^m(\widehat{M_i})$



Form a complete N-bond basis



Impose invariance w.r.t. rotations and permutations





# Sampling the Loss Function Potential

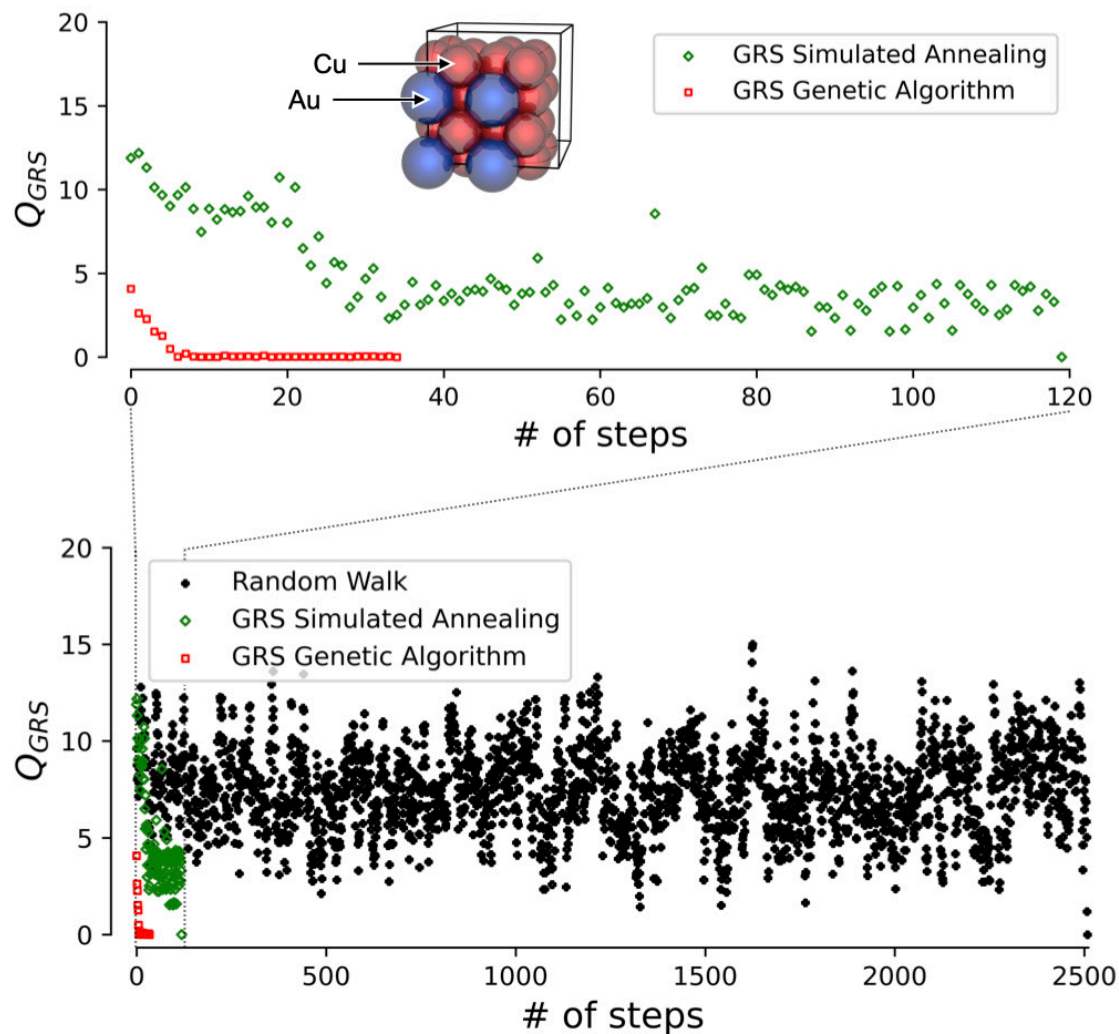
"Generalized representative structures  
for atomistic systems"

OPEN ACCESS

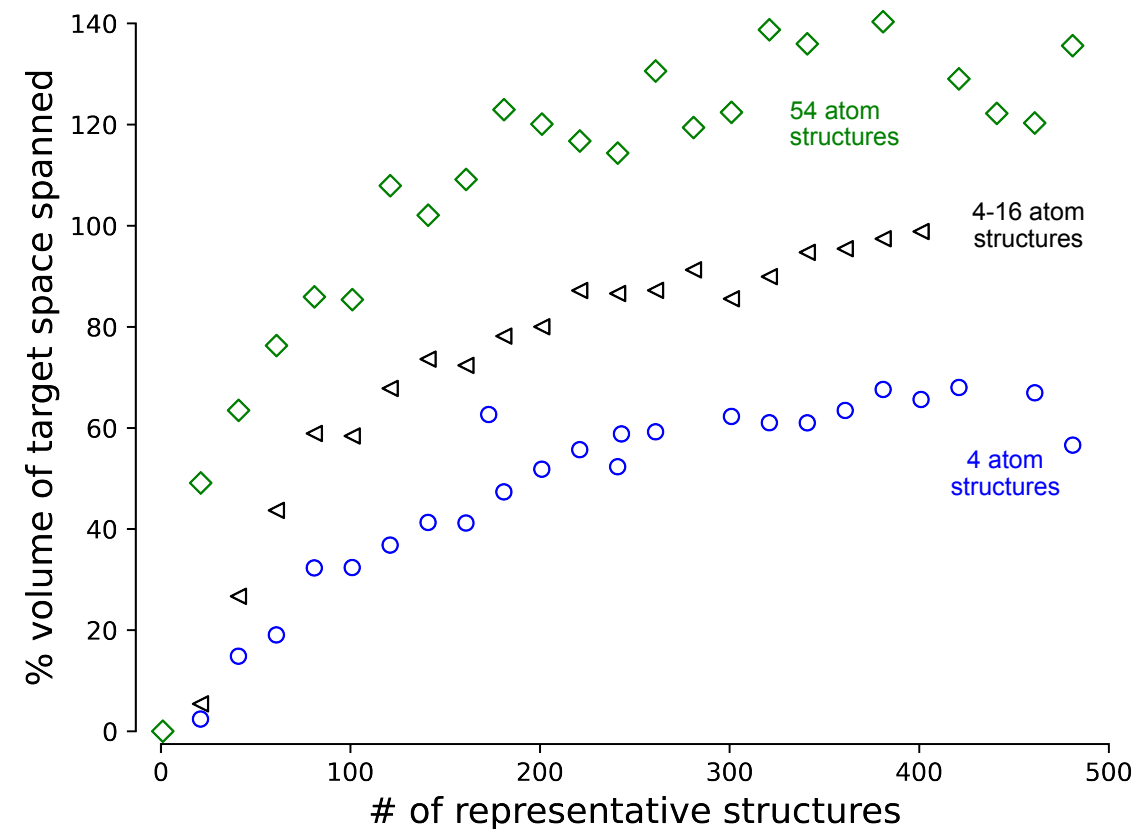
IOP Publishing

Journal of Physics: Condensed Matter

J. Phys.: Condens. Matter **37** (2025) 075901 (13pp) <https://doi.org/10.1088/1361-648X/ad9791>



- More representation in larger structures, batches of structures.

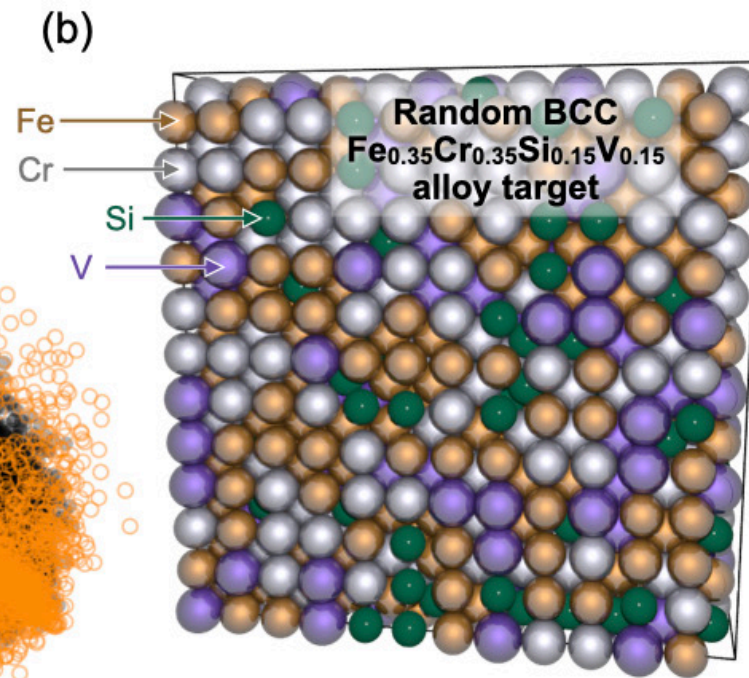
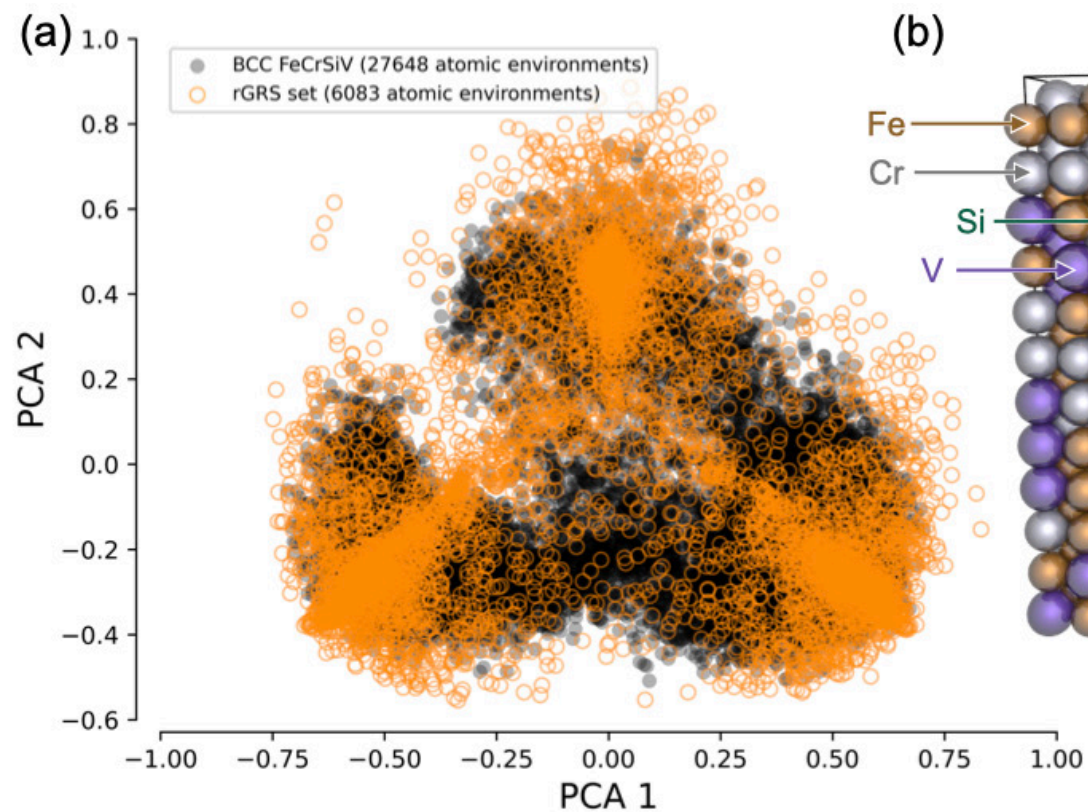






# Representing Complex, Heterogeneous Structures

- Descriptor basis is chemically informed, making complex chemistry and structural disorder capable of miniaturizing.



$$Q_{GRS}^{intra} : 0.036$$

$$\#GRS : 40$$



$$Q_{GRS}^{intra} : 0.038$$

$$\#GRS : 80$$

